

PURIFICATION OF LITHIUM-ION BATTERY BLACK MASS THROUGH TAILORED ALKALINE CORROSION

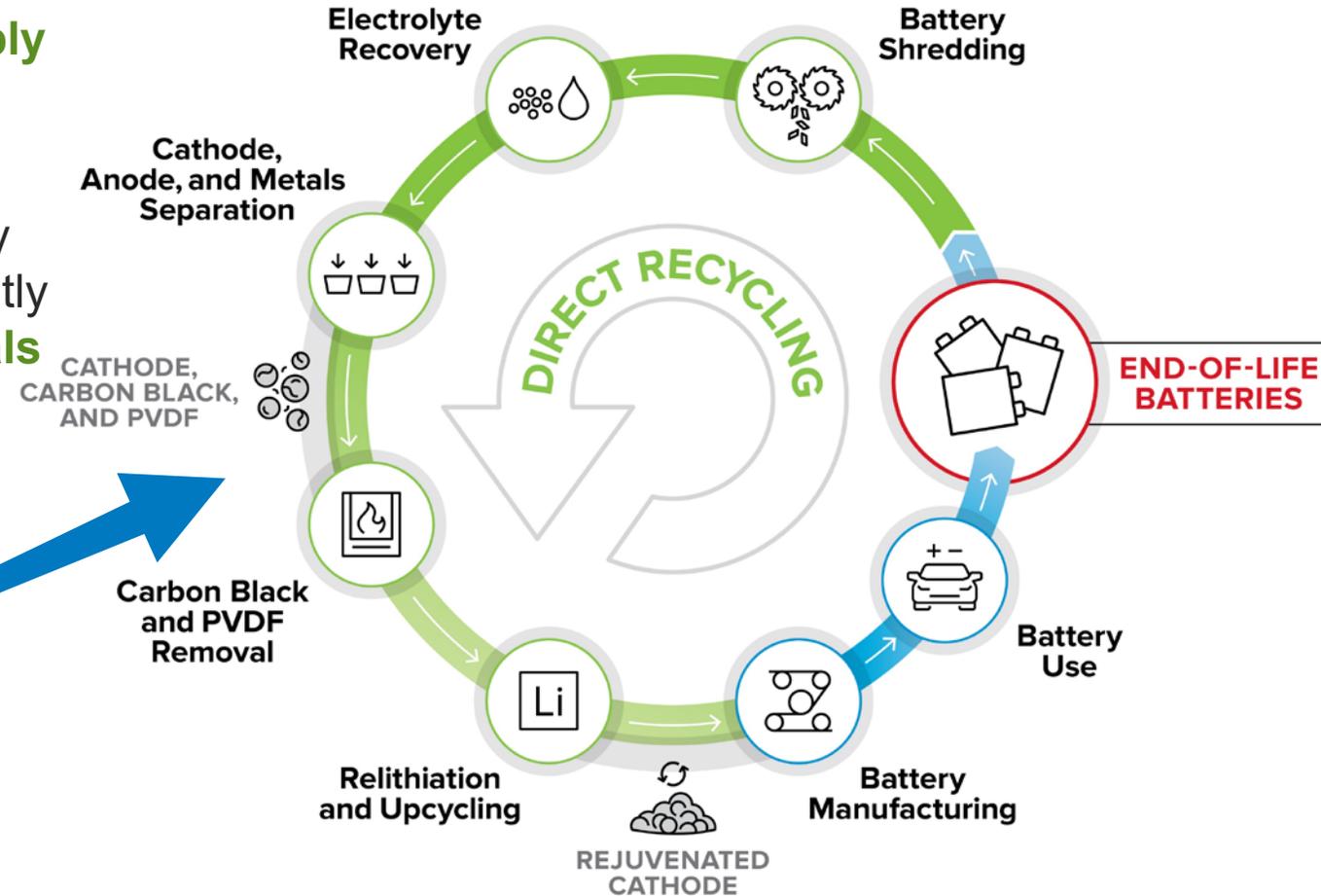
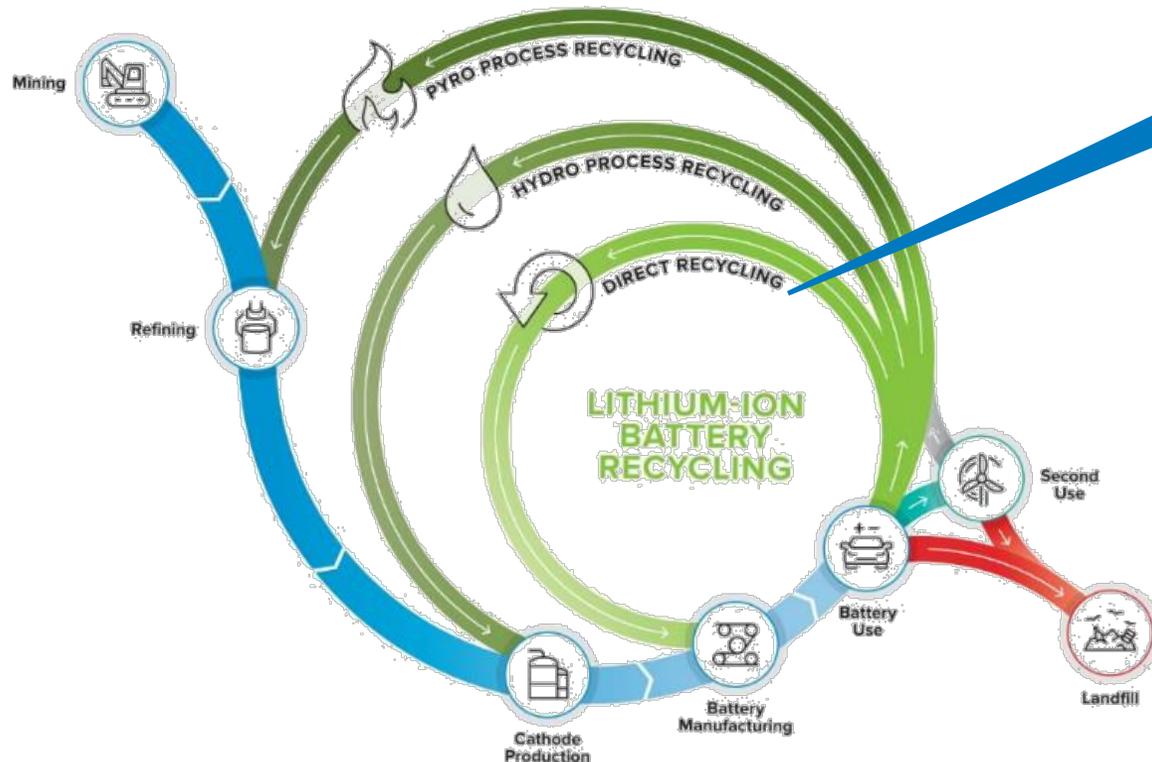


**KAE FINK, PAUL GASPER, MAX SCHULZE, RYAN BROW,
JOSHUA MAJOR, ANDREW COLCLASURE, MATTHEW KEYSER**

National Renewable Energy Laboratory

Direct Recycling of Li-Ion Batteries

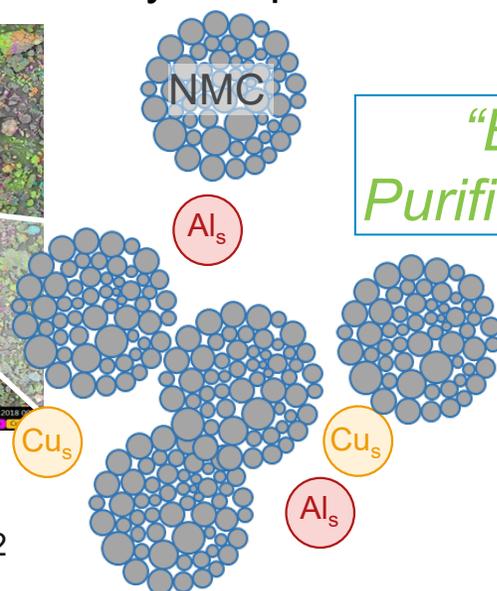
- Burgeoning demand for Li-ion batteries induces **supply chain instability** and raises concerns regarding **end-of-life disposal**.
- DOE goal: “Reduce the cost of electric vehicle battery packs to <\$150/kWh with technologies that significantly reduce or eliminate **dependency on critical materials** and utilize **recycled material feedstocks**”



Direct recycling retains the engineered value of battery materials and minimizes processing steps.

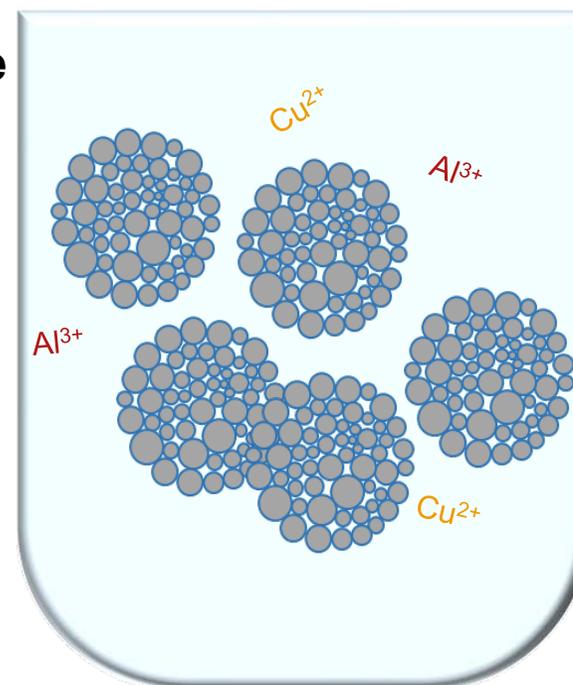
Black Mass Purification: Process Overview

Shredded black mass contains trace **Al** & **Cu** from current collectors that may **inhibit cell performance** and impacts **purity** of recycled product.



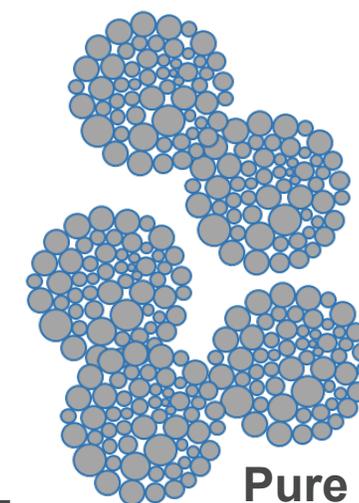
“Contaminated”
 $Li(Ni_xMn_yCo_{1-x-y})O_2$
(NMC)
 $NMC + Al^0, Cu^0$

“BM
Purification”



$NMC + Ionized$
 Al^{3+}, Cu^{2+}

Physical
Filtration



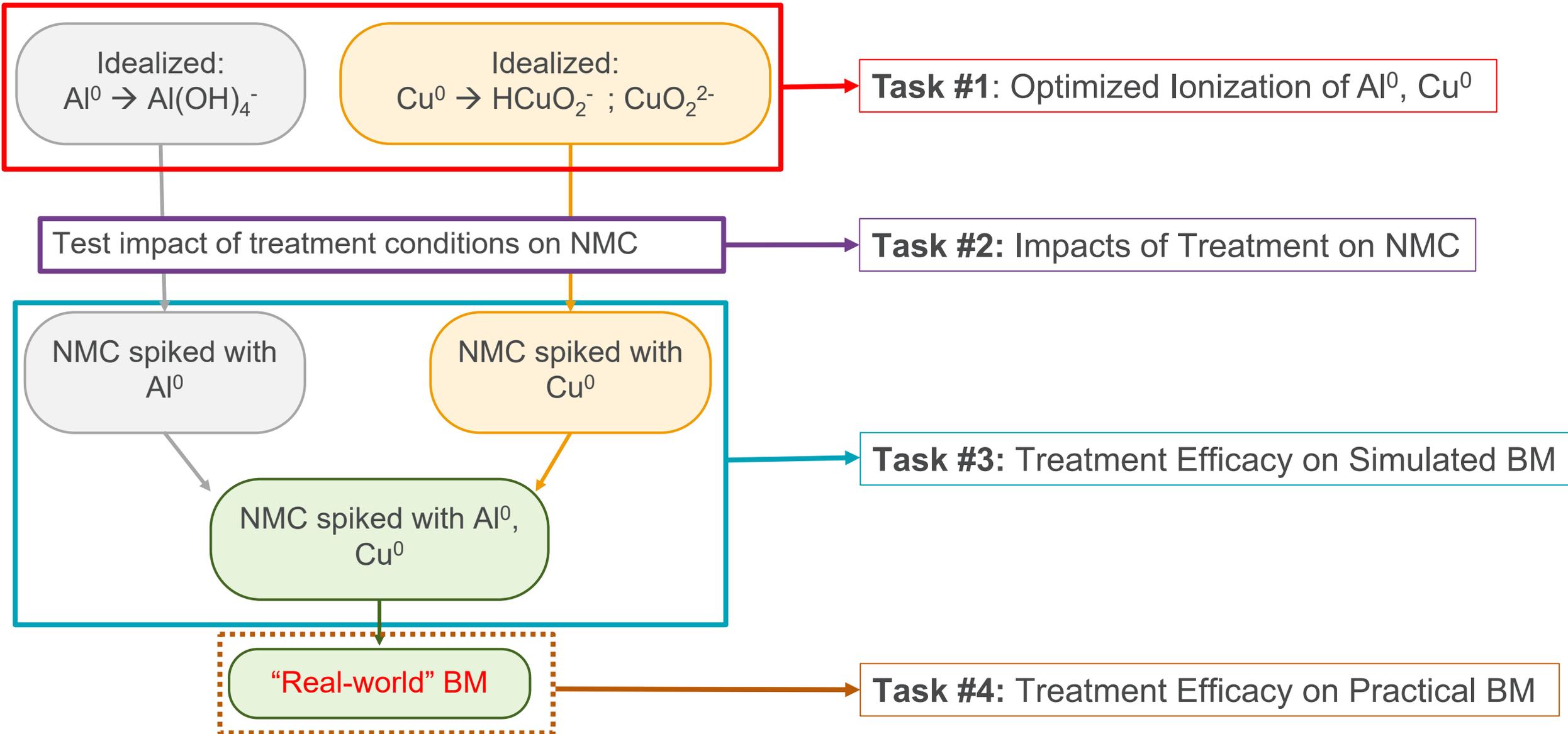
Pure NMC

+
 Cu_s Al_s
 Cu_s Al_s
Pure Recovered Al^0, Cu^0

Goal: Identify and optimize BM purification process to enable **complete and rapid dissolution** of solid contaminants (Al^0, Cu^0) **without adversely impacting** structure or electrochemical performance of NMC.

Approach: “Kinetically & thermodynamically assisted” **alkaline aqueous corrosion**

Overview of Project Workflow

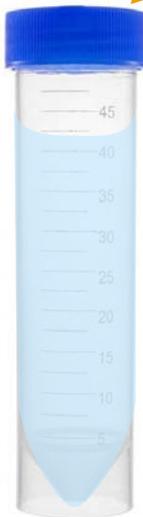


Task #1: Optimized Ionization of Al_0 , Cu_0

Idealized:
 $\text{Al}_0 \rightarrow \text{Al}(\text{OH})_4^-$

Idealized:
 $\text{Cu}_0 \rightarrow \text{HCuO}_2^- ; \text{CuO}_2^{2-}$

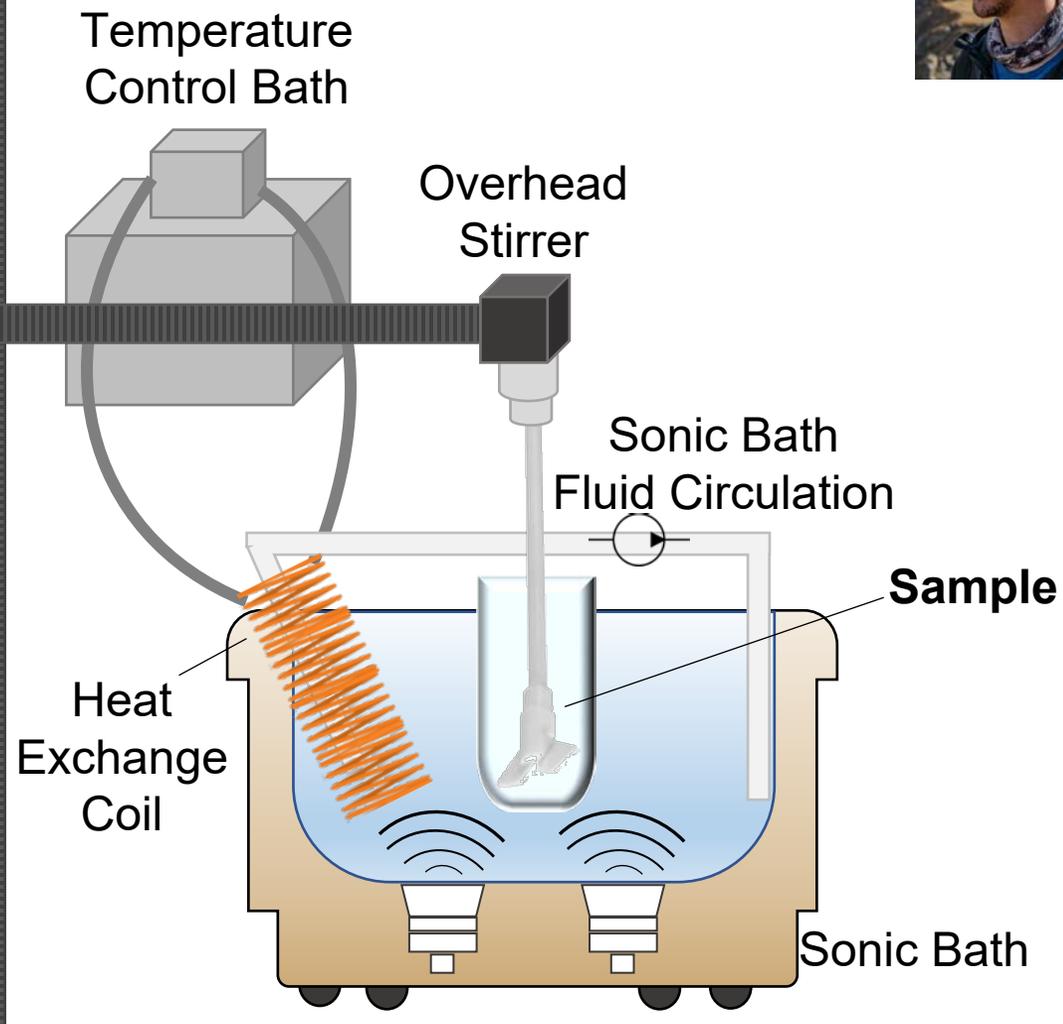
Al^0 or Cu^0 (~1/3 of RT solubility)
DI H_2O →
 OH^- (conc.; "carbonate-free")
 Cl^- (saturated; ~4.5 M)
DTPA (2x molar equivalent)



"Bench-scale" testing:
• 40-45 mL solution
• 10 mg contaminant

Modular overhead sonic-stirrer

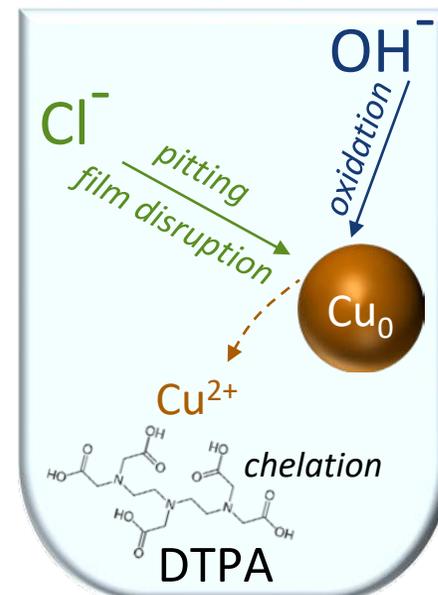
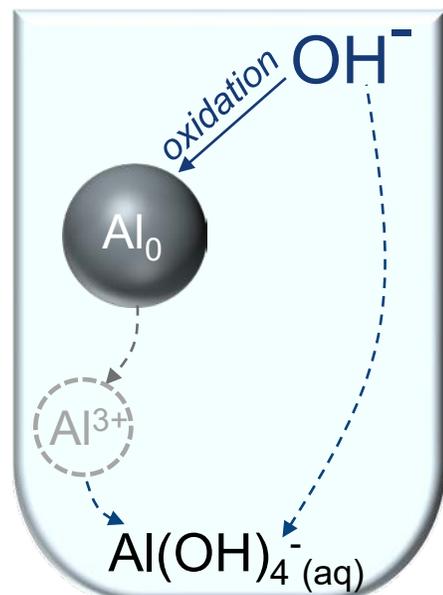
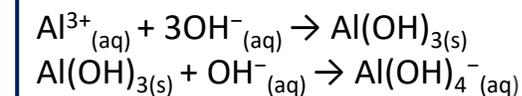
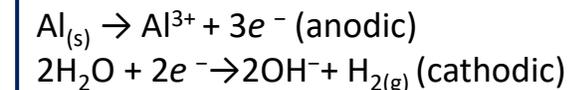
Designed and built support: Joshua Major (NREL)



Al⁰ & Cu⁰ Corrosion: Theoretical Foundations

Al⁰ Corrosion:

-Rapid, strongly pH-dependent reaction under ambient oxidative conditions

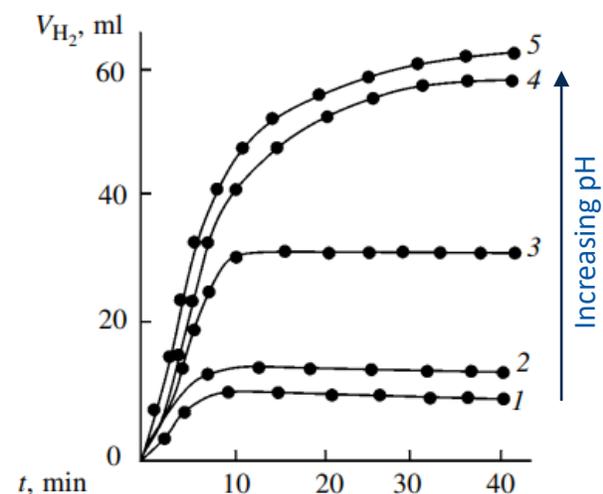


Cu⁰ Corrosion:

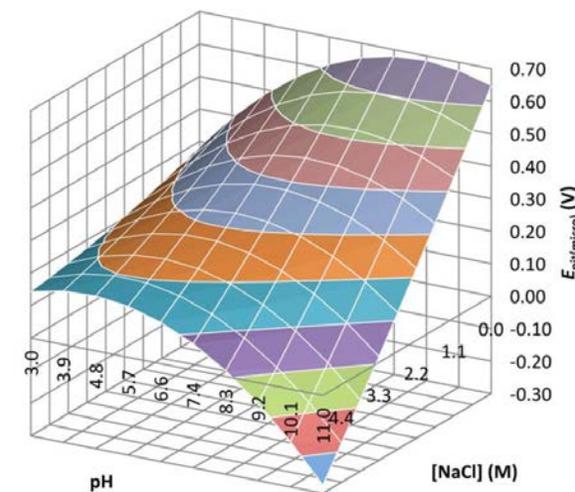
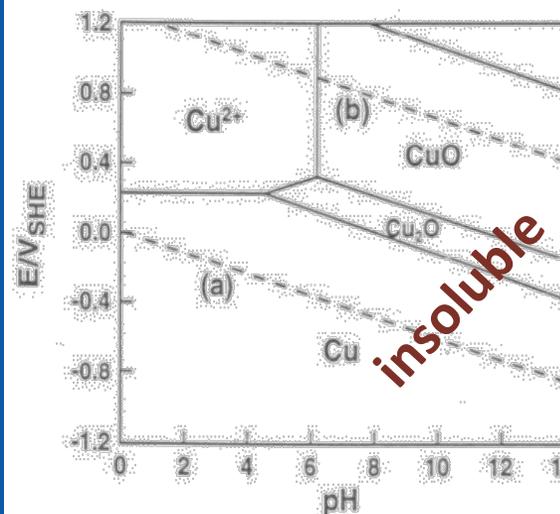
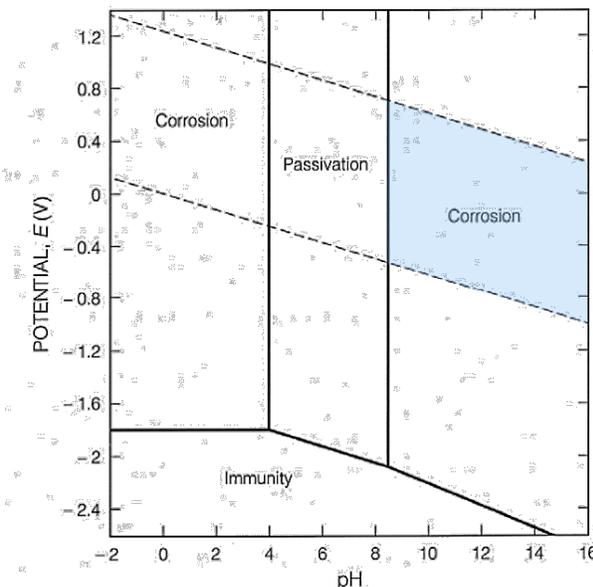
-Kinetically slow and thermodynamically unfavorable

-Low solubility of Cu²⁺ species and formation of passivating surface films

-**Chloride** reportedly enhances corrosion in alkaline media through pitting and surface film disruption



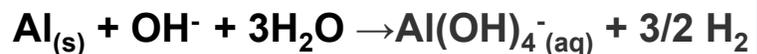
Alexsandrov et al., Russian J. Gen. Chem. 73, 5 (2003), 689-694.



Arjmand and Adriaens, Materials 2012, 5, 2439-2464

Al⁰ Corrosion: Bench-Scale Optimization

Approach: pH monitoring (*unbuffered system*)



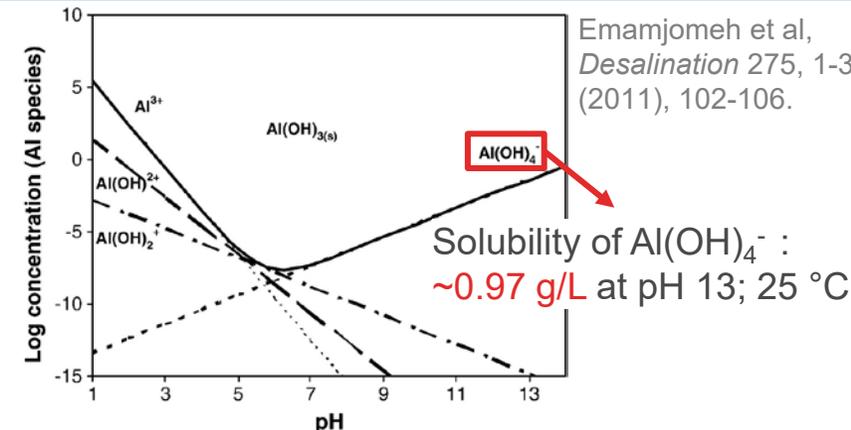
Each extent of reaction consumes Al₀ and OH⁻ on a 1:1 molar basis

Reaction kinetics and extent may be quantified by **tracking solution pH**

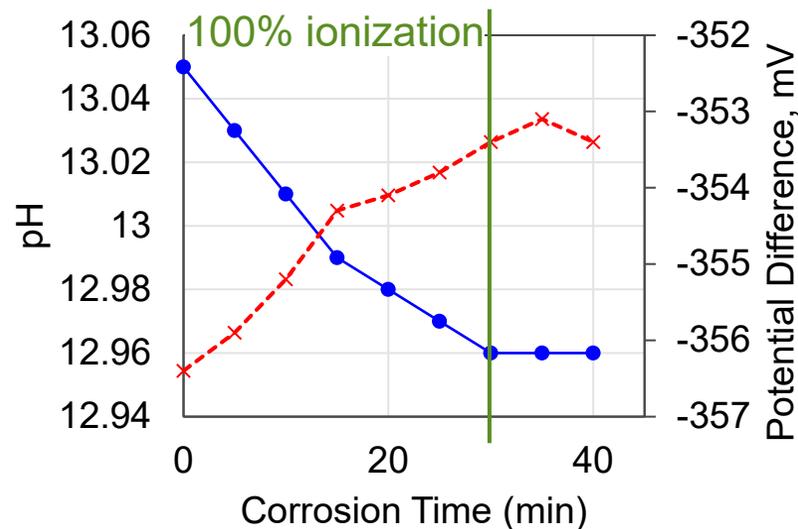
$$\log K_f [\text{Al}(\text{OH})_4^-] = 33.0 \text{ (25 } ^\circ\text{C)}$$

Al³⁺ preferentially and strongly binds OH⁻ in alkaline solution to form soluble Al(OH)₄⁻.

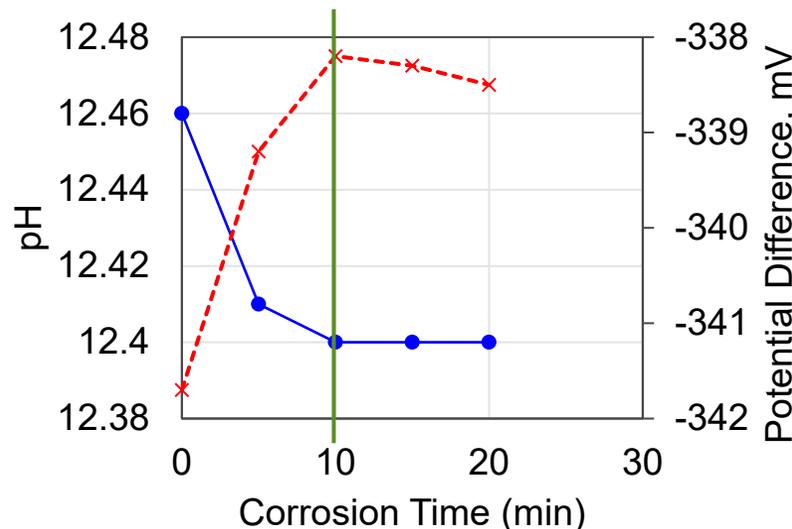
Thermodynamically favorable reaction;
room for kinetic tuning.



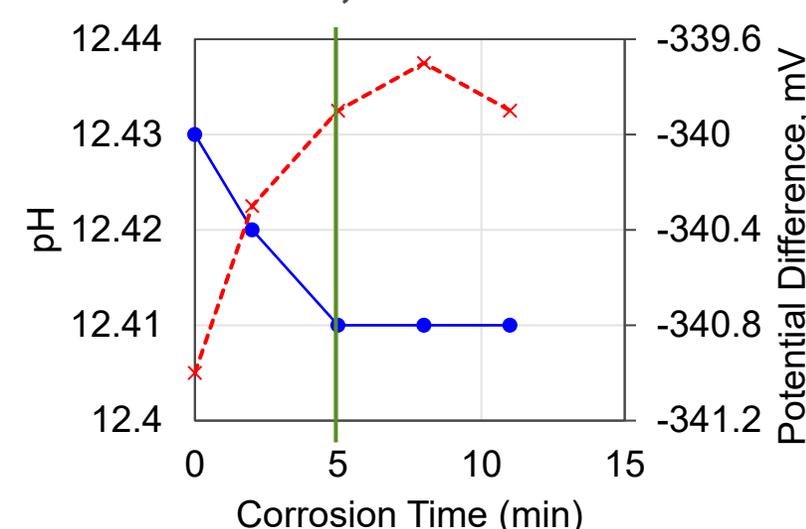
RT, no T control (20–22 °C)



40 °C, No Sonics



40 °C, With Sonics



All samples were prepared at pH 13 at calibrated room temperature (~22 °C)

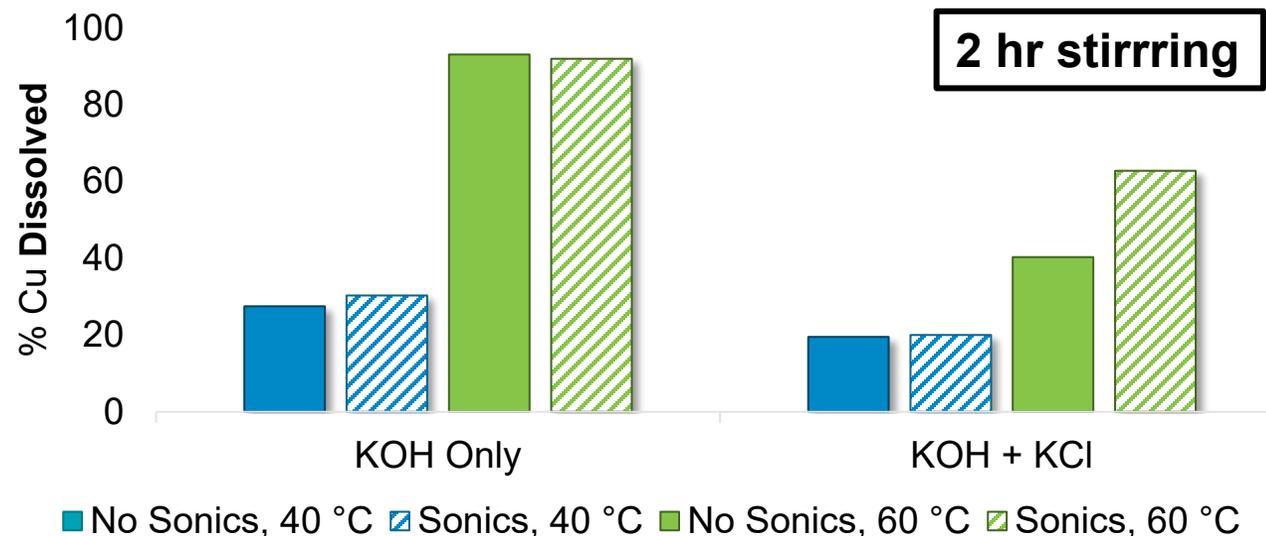
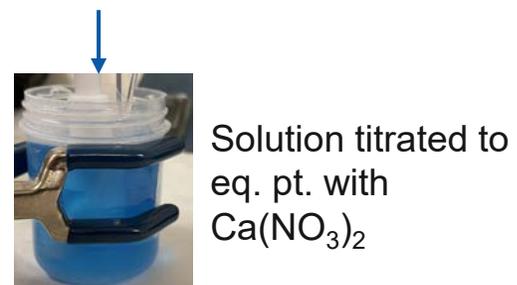
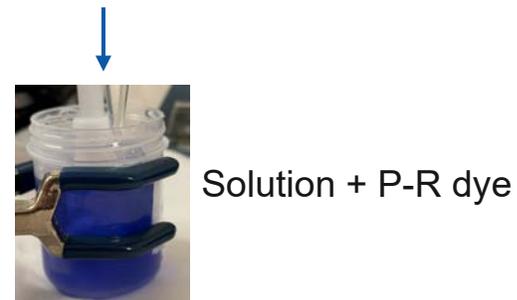
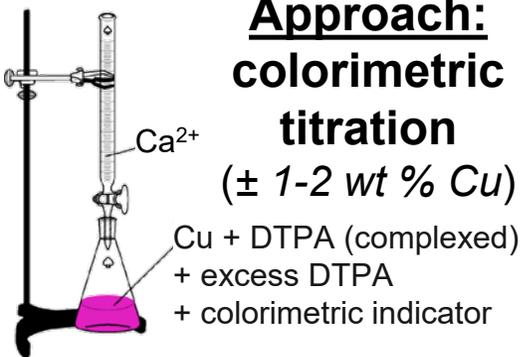
Mildly elevated temperature and sonication increase corrosion rate for Al⁰.
Full corrosion achieved in **5 min** (40 °C with sonication; 10 mg Al⁰ at pH 13)

Cu⁰ Corrosion: Bench-Scale Optimization

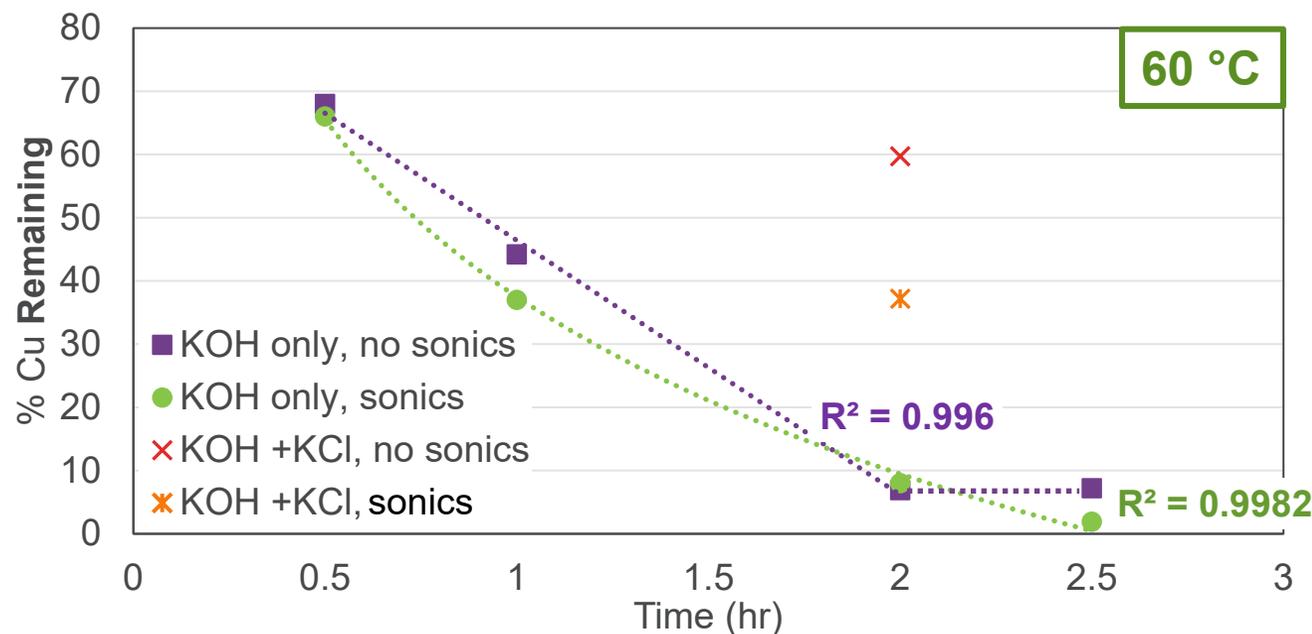
All samples shown were prepared at pH 13 (measured at RT)

Approach: colorimetric titration

(± 1-2 wt % Cu)



- Cl⁻ *inhibits* Cu⁰ corrosion at moderate temperatures.
- Sonication improves extent of ionization for Cl⁻-treated samples, but does not sufficiently compensate for passivating effects.



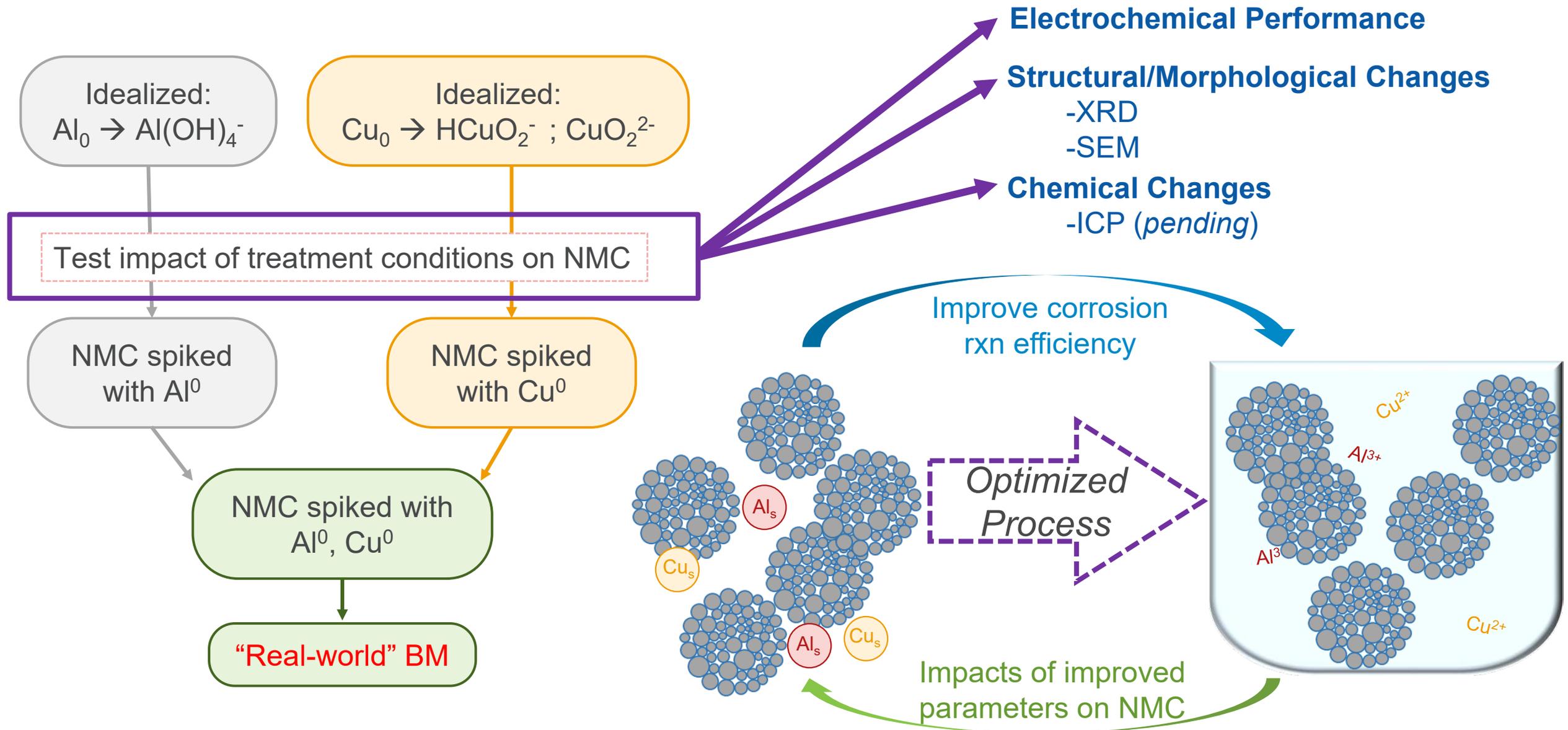
Without Sonication:

- **Linear (0th order)** kinetics until ~93% ionization
- Thermodynamic maximum <100% ionization

With Sonication:

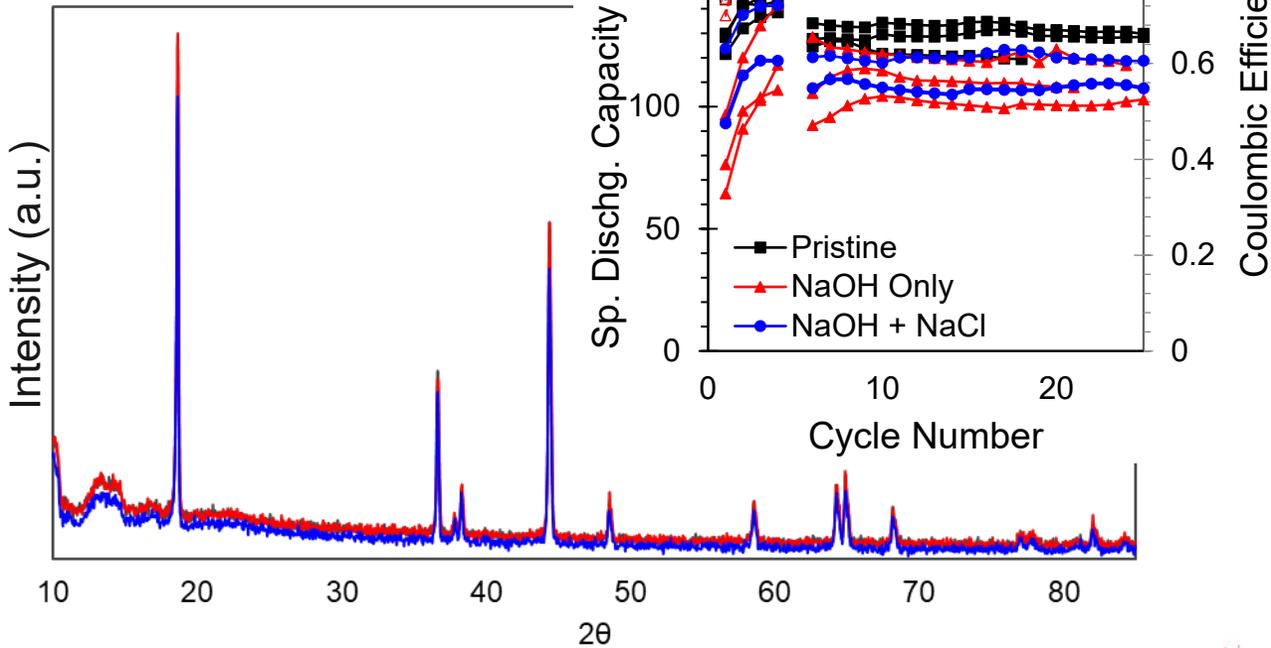
- **Logarithmic (1st order)** kinetics (conc. dependence on ionized product)
- ~100% corrosion achieved after **2.5 hr** (*within error*)

Task #2: Impacts of Treatment on NMC

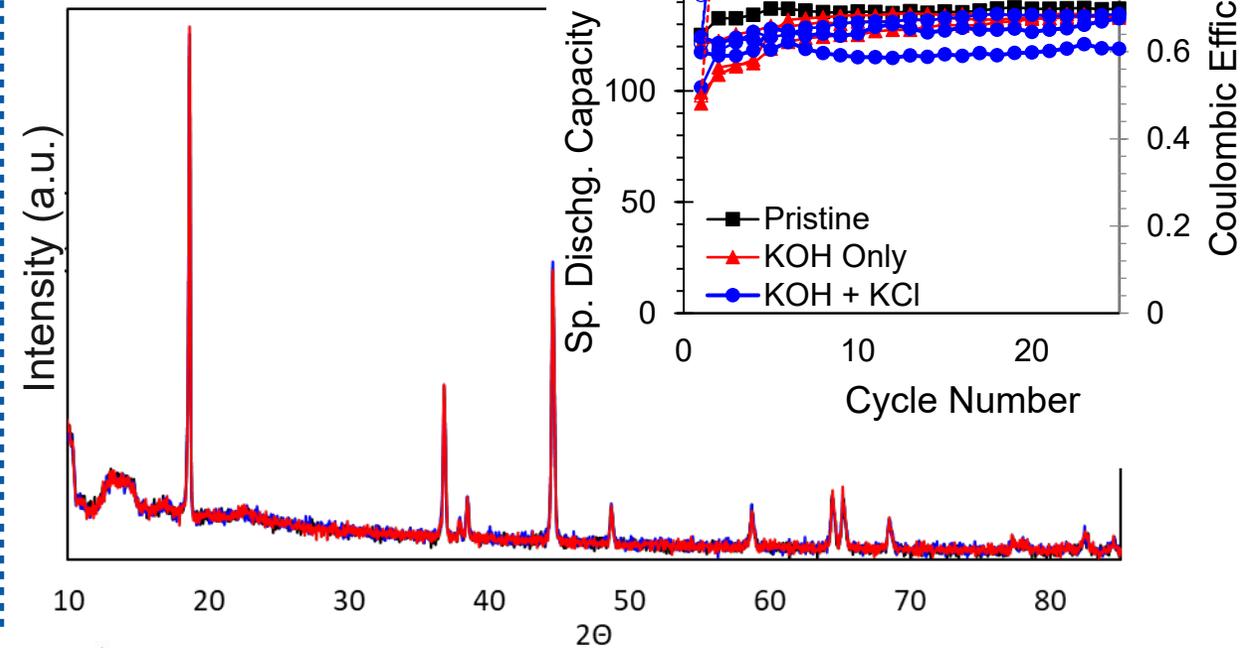


Iterative Process Optimization

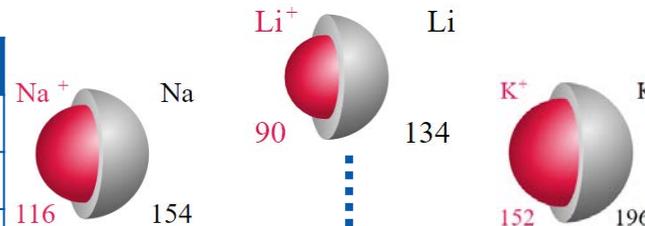
NMC-532; Na⁺ salts pH 11



NMC-532; K⁺ salts pH 13



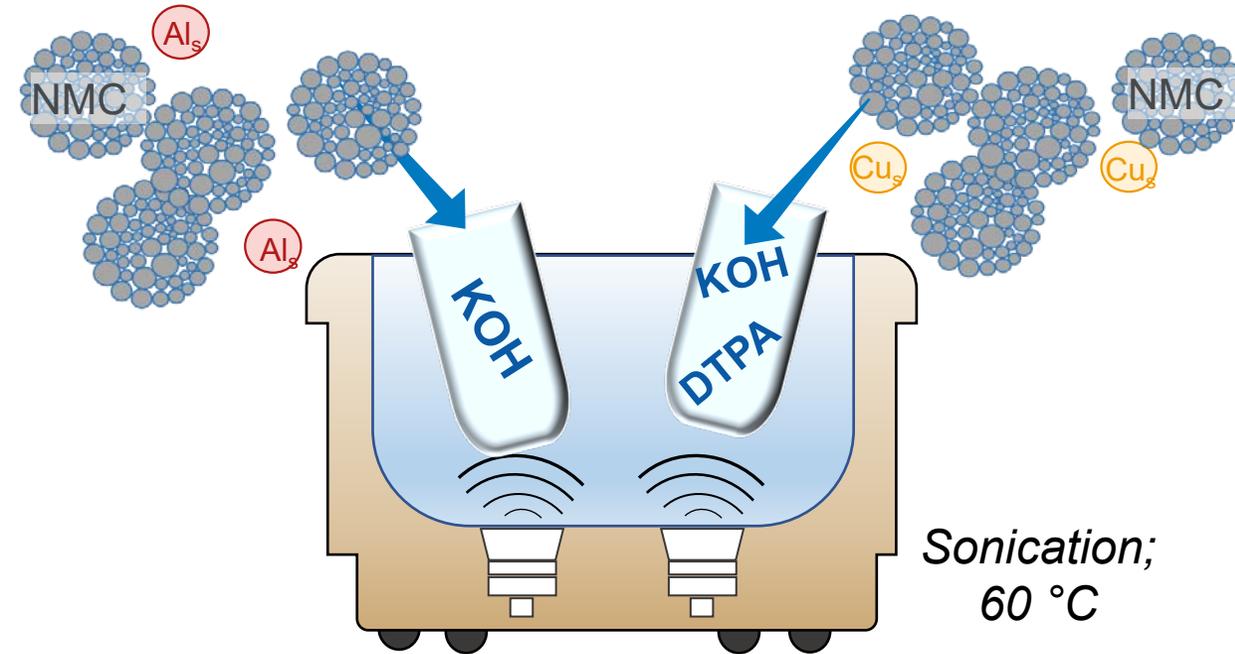
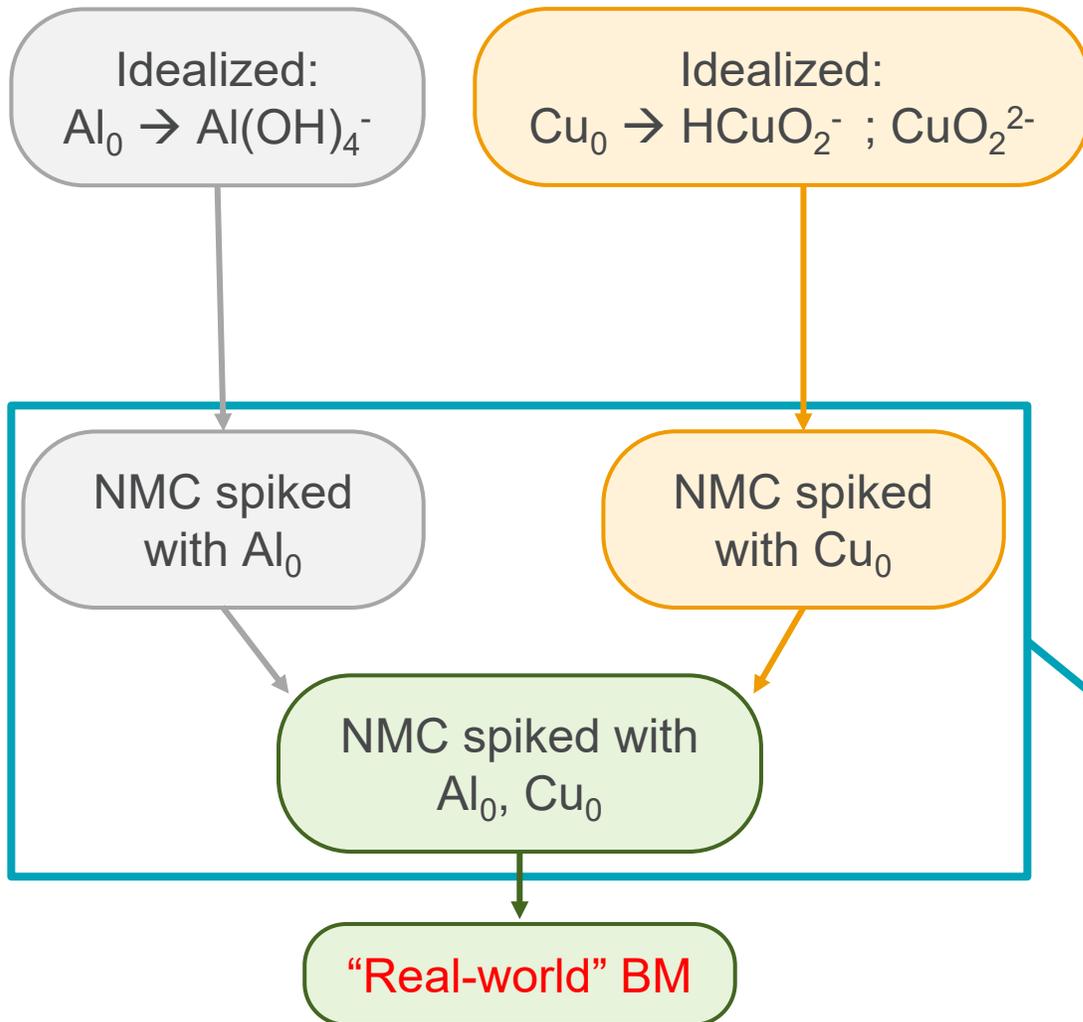
Condition	% Co ₃ O ₄ ± SE	a	c	I(003)/I(104)
Pristine	1.0 ± 0.7	0.288	1.426	1.03
NaOH	1.5 ± 0.9	0.289	1.425	1.06
NaOH/NaCl	1.1 ± 0.5	0.288	1.426	1.12



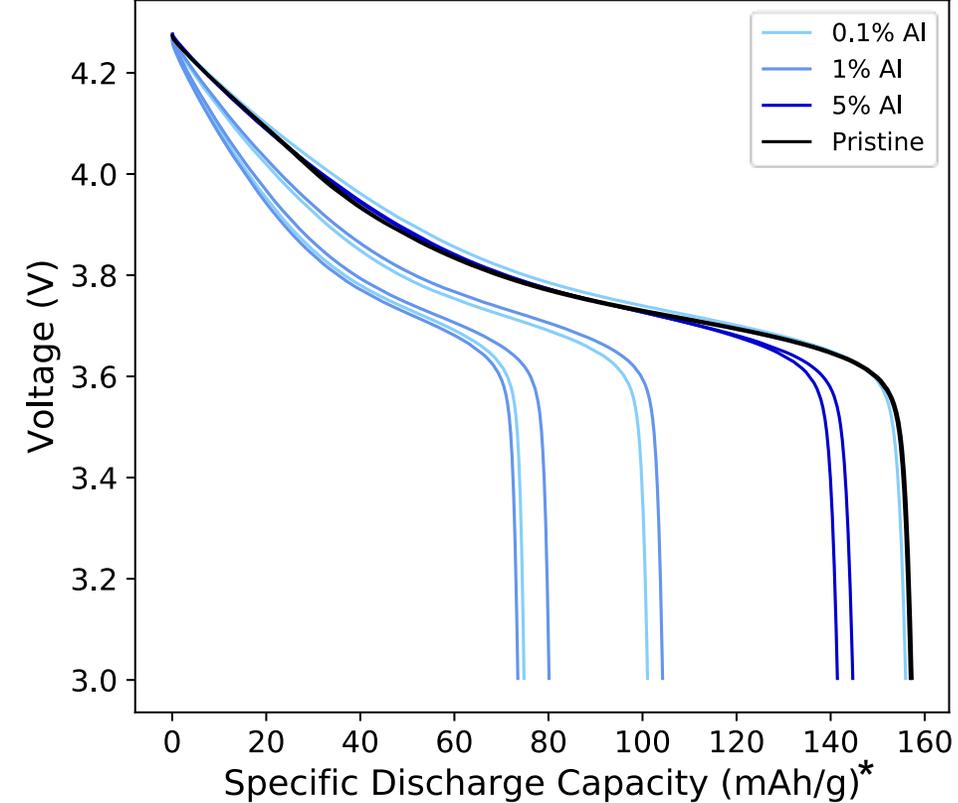
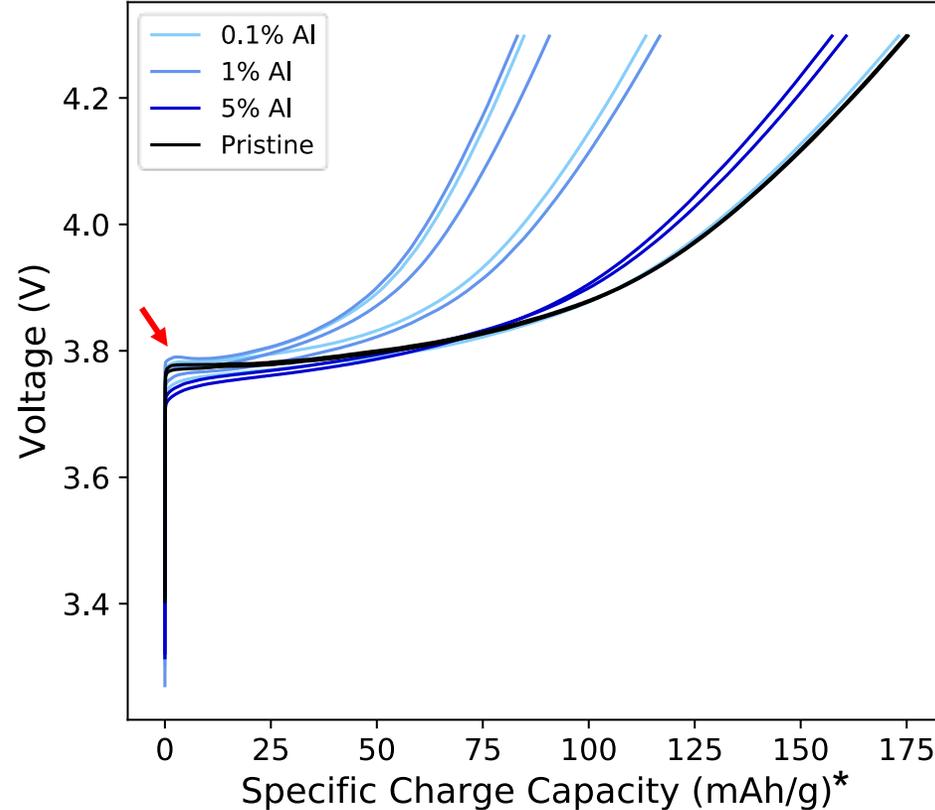
Condition	% Co ₃ O ₄ ± SE	a	c	I(003)/I(104)
Pristine	0.05 ± 0.01	0.286	1.423	1.05
KOH	0.04 ± 0.01	0.286	1.423	1.08
KOH/KCl	0.05 ± 0.01	0.286	1.422	1.09

Evidence of cation mixing and bulk structural rearrangement with Na⁺ salts prompted shift to K⁺ salts.

Task #3: Treatment Efficacy on Simulated BM



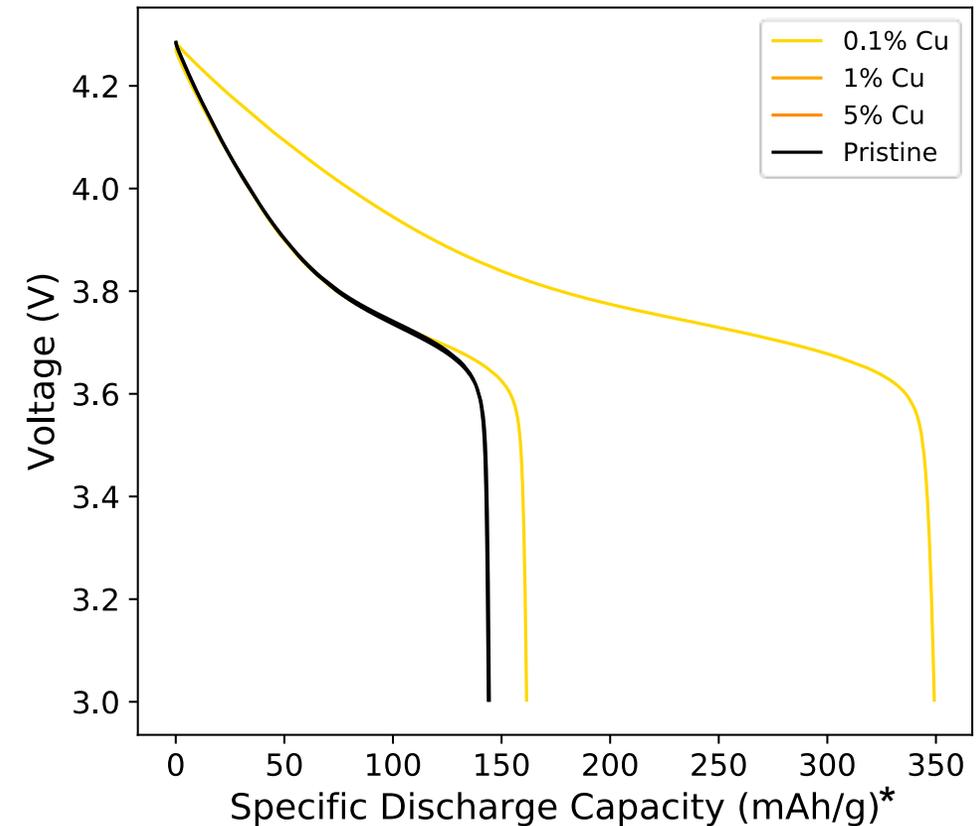
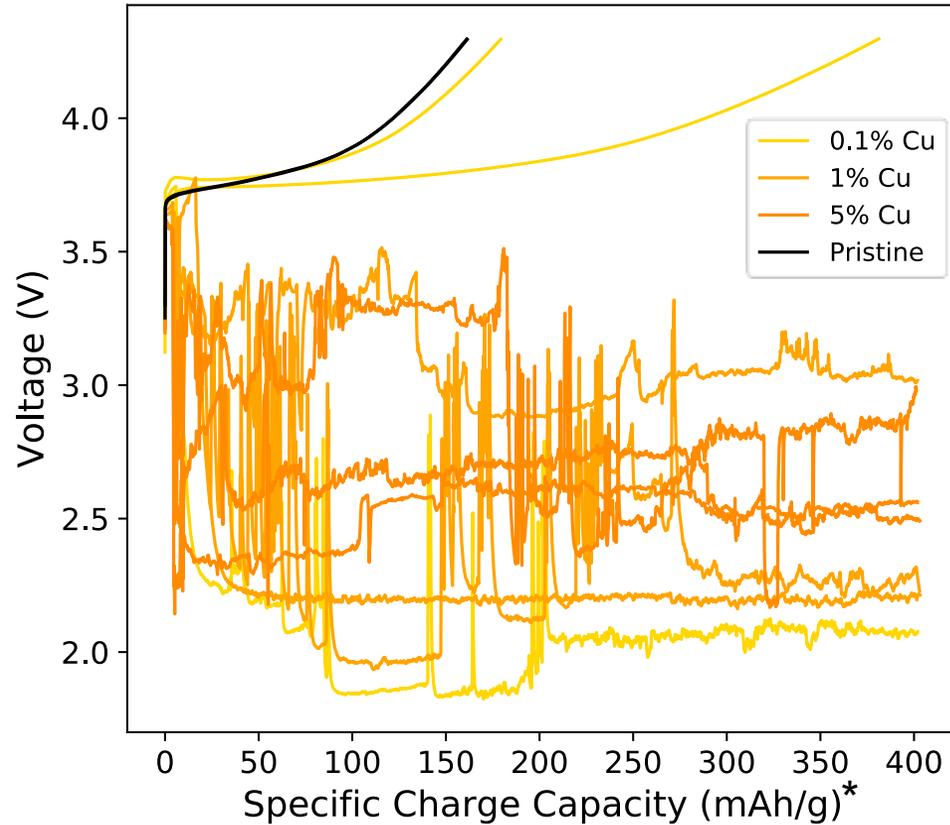
Prepare batches of NMC spiked with relevant contaminant loadings (0.1 – 5 wt%)
Electrochemical & structural testing on spiked NMC before and after BMP treatment



*Specific capacity normalized with respect to mass of NMC (contaminant mass excluded)

- Al⁰ contamination may function as a conductive dopant (reduces initial charge resistance), particularly at higher impurity concentrations,¹ but negatively impacts overall cell capacity (higher overpotential)
- Interestingly, the worst impact on performance is seen between 0.1% - 1% Al⁰...i.e., the practically relevant contamination level for shredded black mass

¹Fink, K. et al., Influence of Metallic Contaminants on the Electrochemical and Thermal Behavior of Li-Ion Electrodes (*submitted*)

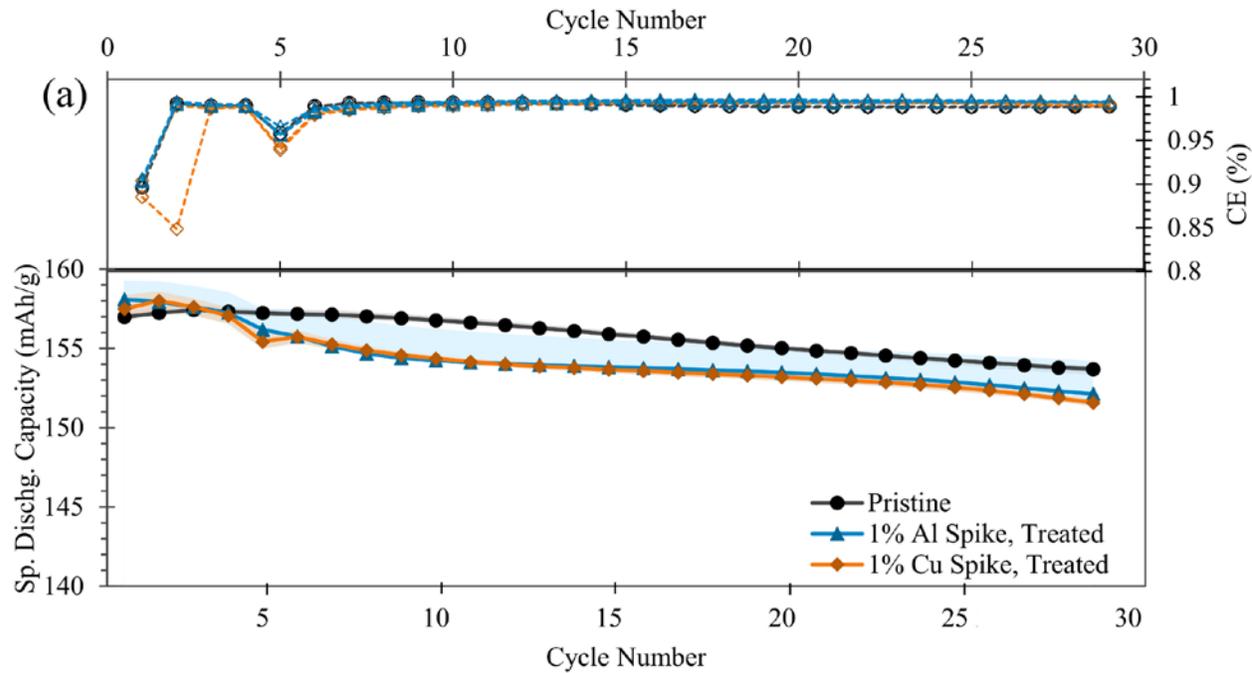


*Specific capacity normalized with respect to mass of NMC (contaminant mass excluded)

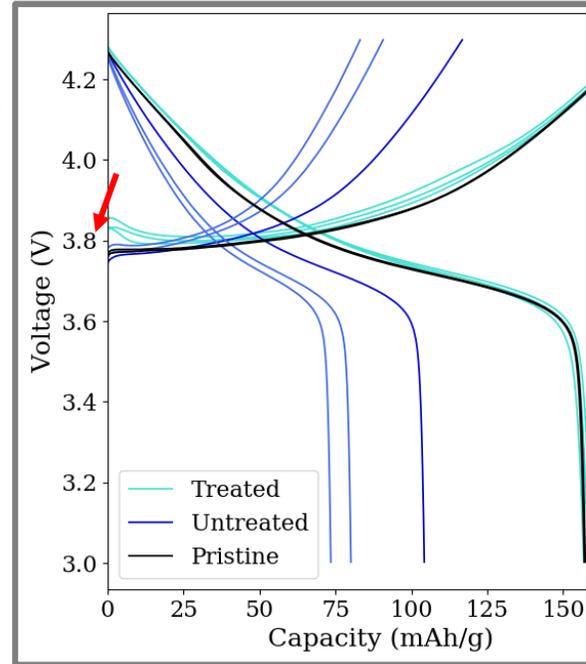
- Practically relevant level of Cu⁰ concentration in black mass (~1%) is detrimental to performance (consistent with previous findings)¹
- Apparent increase in capacity for some 0.1% Cu⁰ replicates attributed to irregular and continued reactivity

¹Fink, K. et al., Influence of Metallic Contaminants on the Electrochemical and Thermal Behavior of Li-Ion Electrodes (*submitted*)

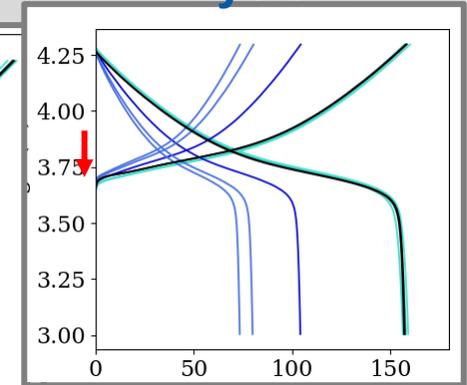
Treatment of Al⁰ and Cu⁰-Contaminated NMC



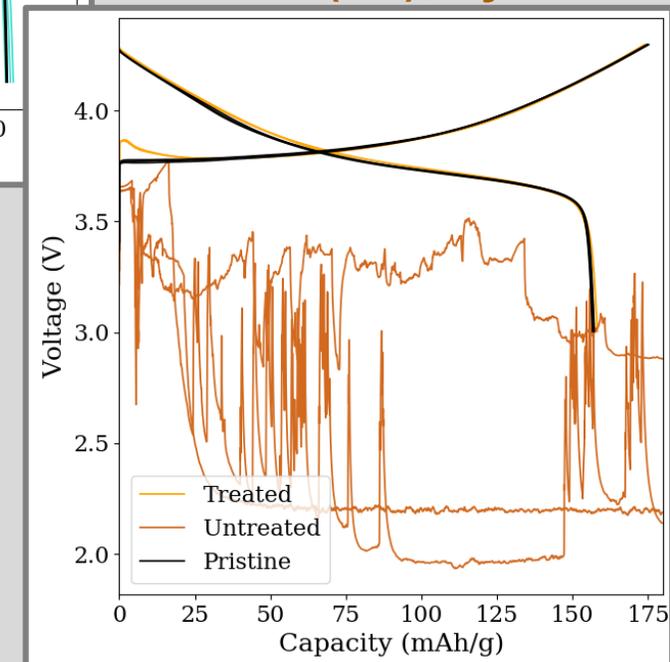
Treated (Al), Cycle 1



Cycle 2

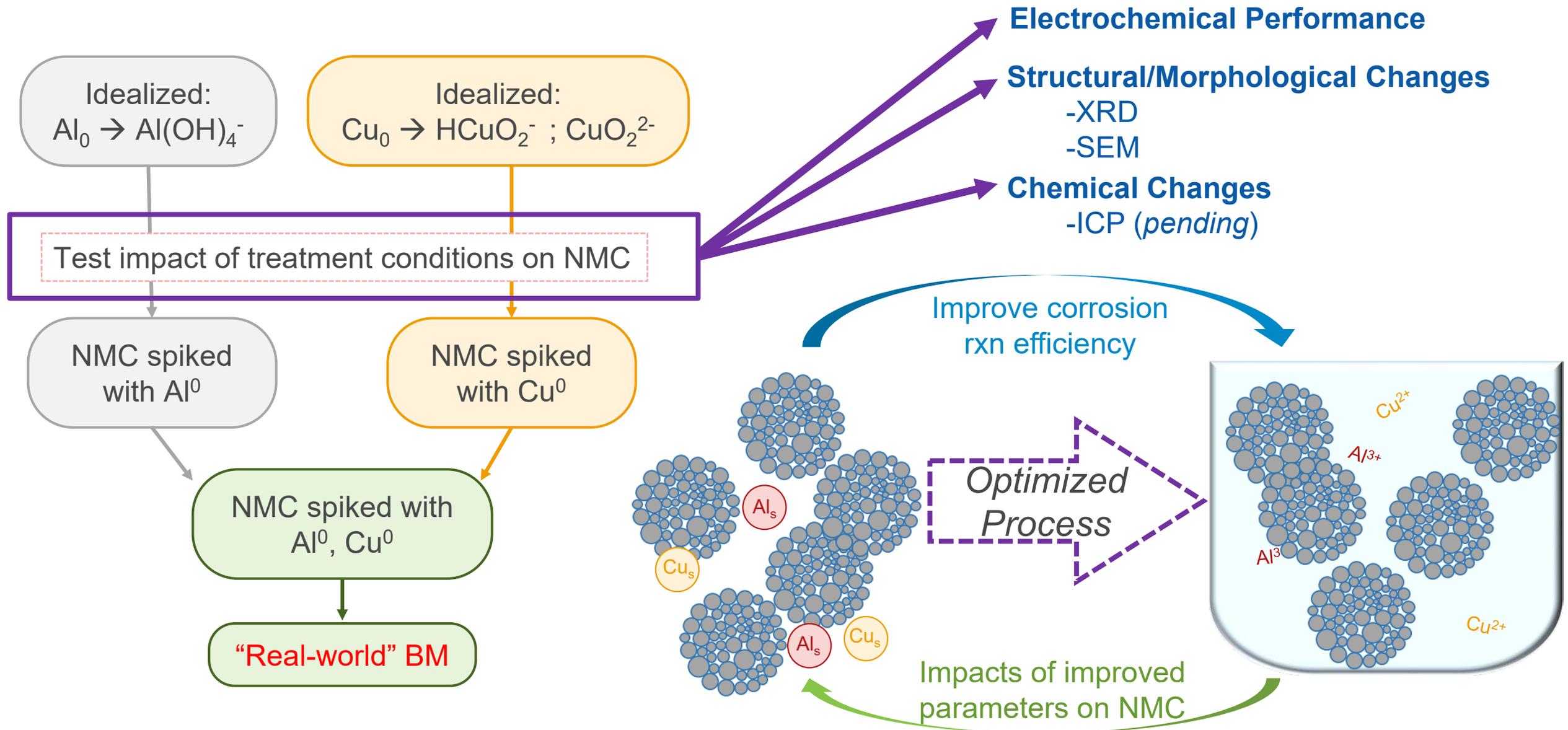


Treated (Cu), Cycle 1

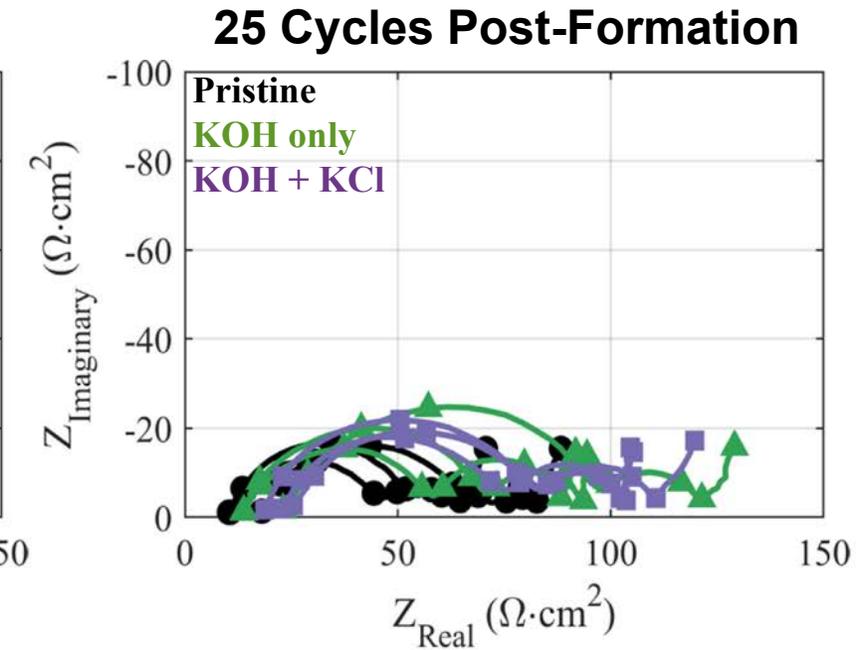
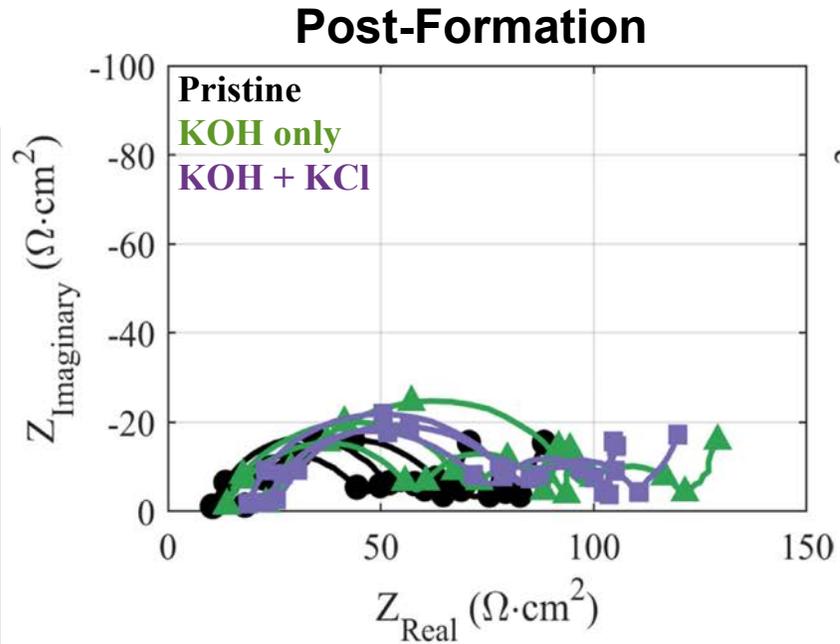
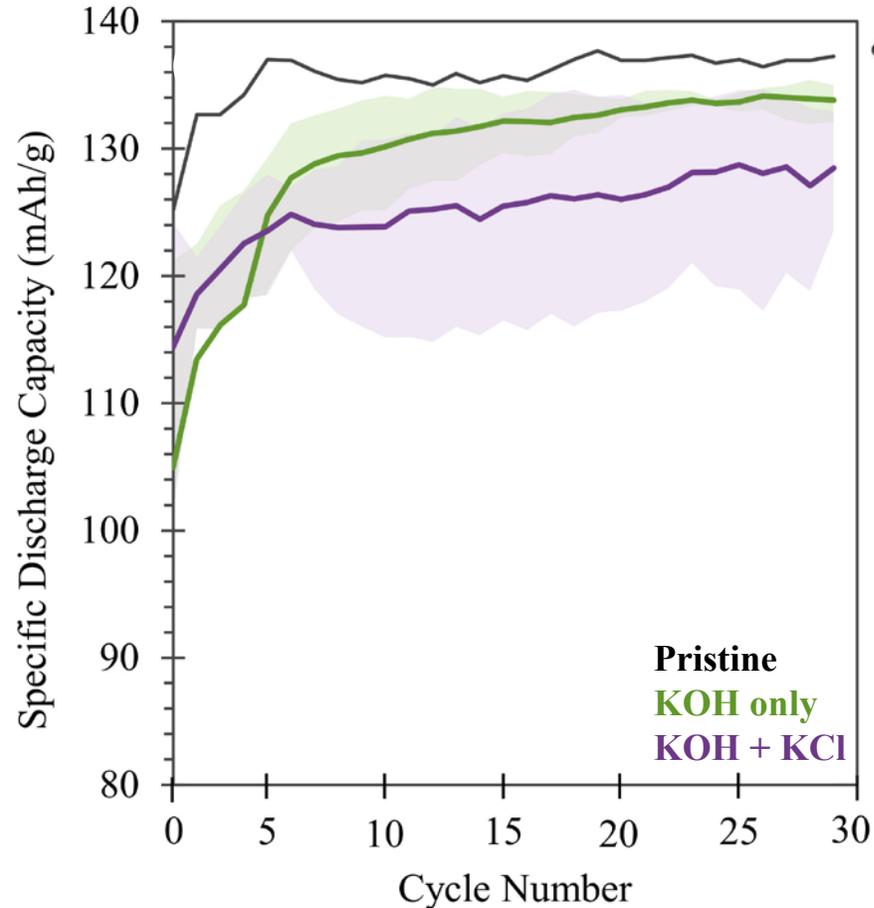


- Half-cell capacity is nearly identical to pristine material, suggesting a **successful purification process that does not adversely impact bulk NMC performance.**
- Irregular high resistance observed at beginning of first cycle; pristine behavior recovered on cycle 2 and beyond.
- This suggests the presence of a reactive surface species that is electrochemically consumed upon initial charge.

Task #2: Impacts of Treatment on NMC (Again)

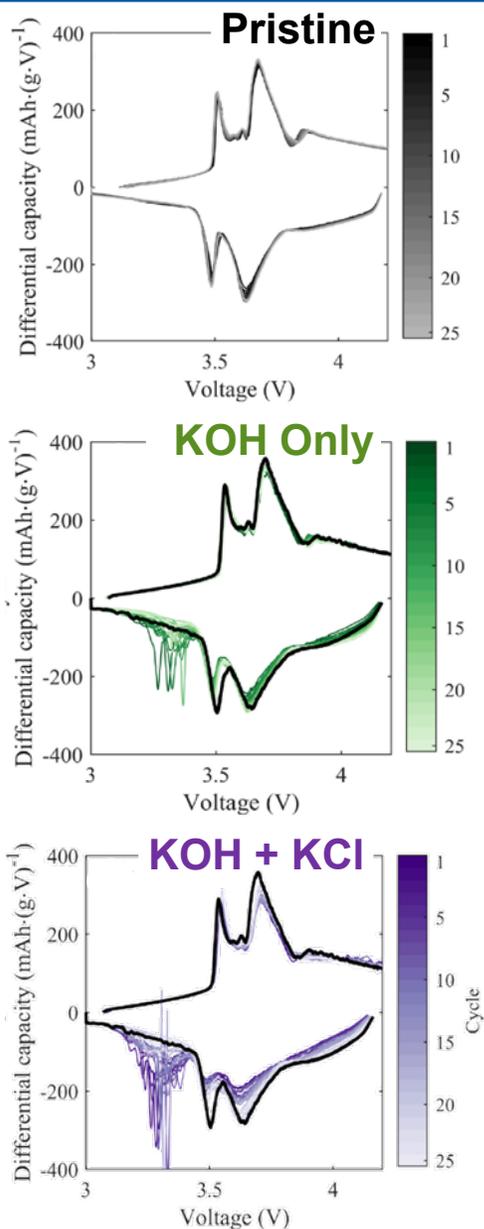


Full-Cell Performance: C/10 Cycling & Impedance

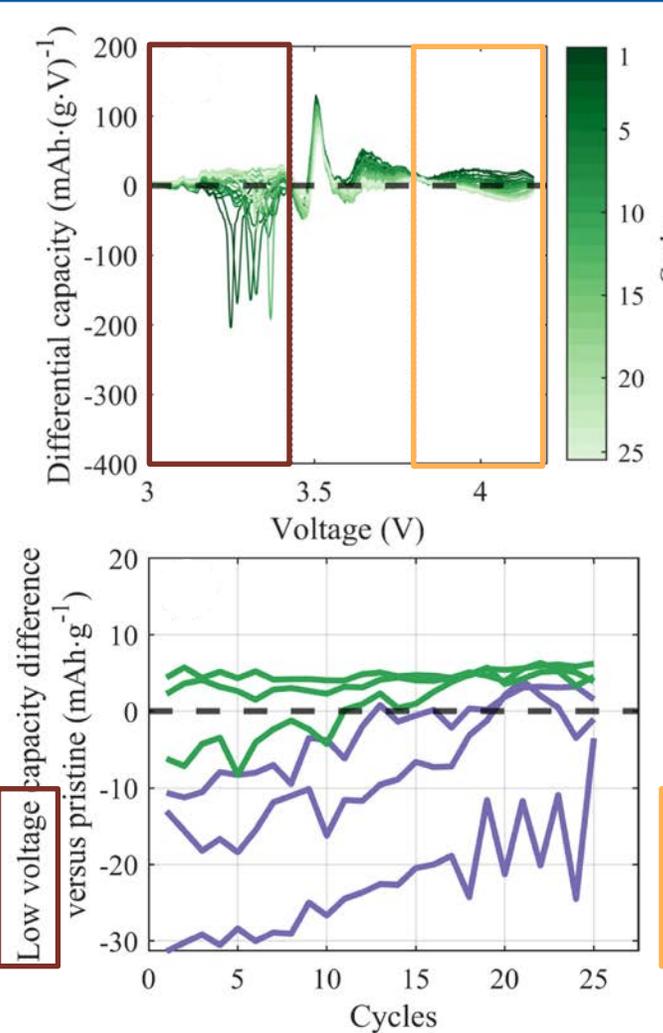


- Capacity reduction & impedance growth observed in full cells (vs pristine graphite) – but not in half cells
- This suggests the presence of *soluble* reactive contaminant(s) in the treated cathode material, which cross over & act adversely at the anode
- Surface contaminants initially prevent effective utilization Li inventory, but do not deplete significant Li through reaction (capacity recovery)

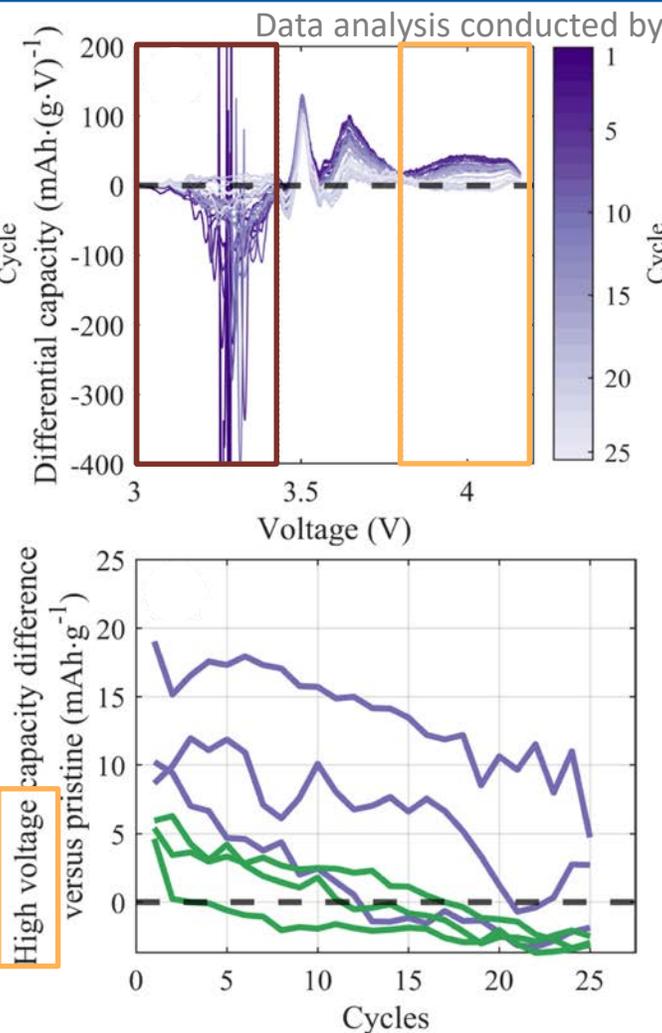
Full-Cell Differential Capacity Analysis



reduced side reactions



Low voltage capacity difference versus pristine ($\text{mAh}\cdot\text{g}^{-1}$)



High voltage capacity difference versus pristine ($\text{mAh}\cdot\text{g}^{-1}$)

recovery of NMC capacity

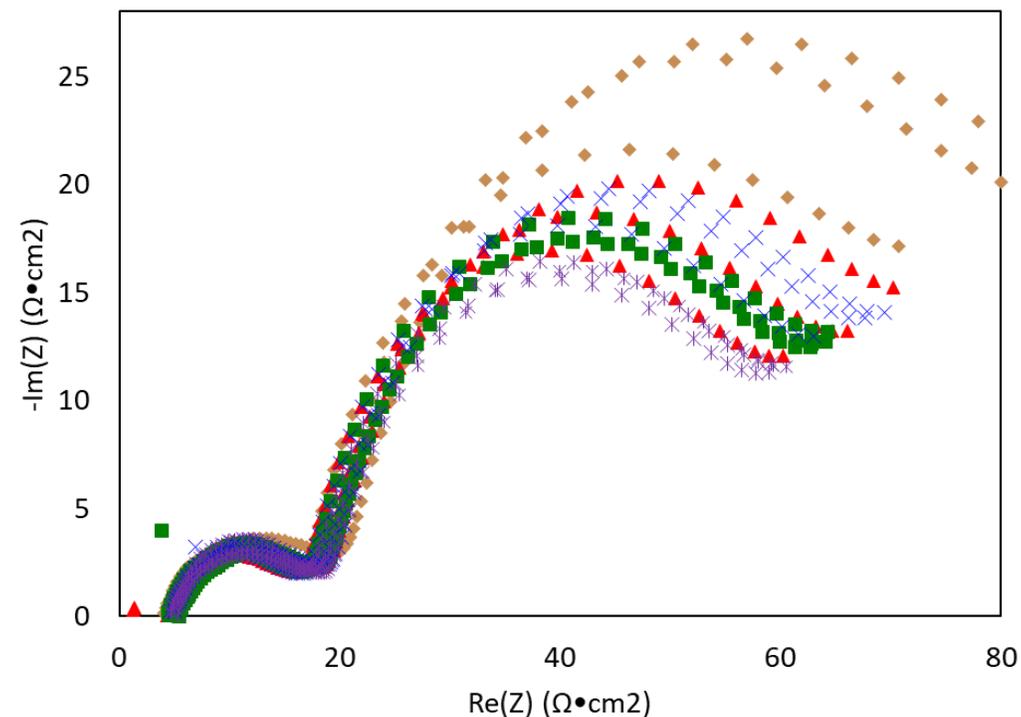
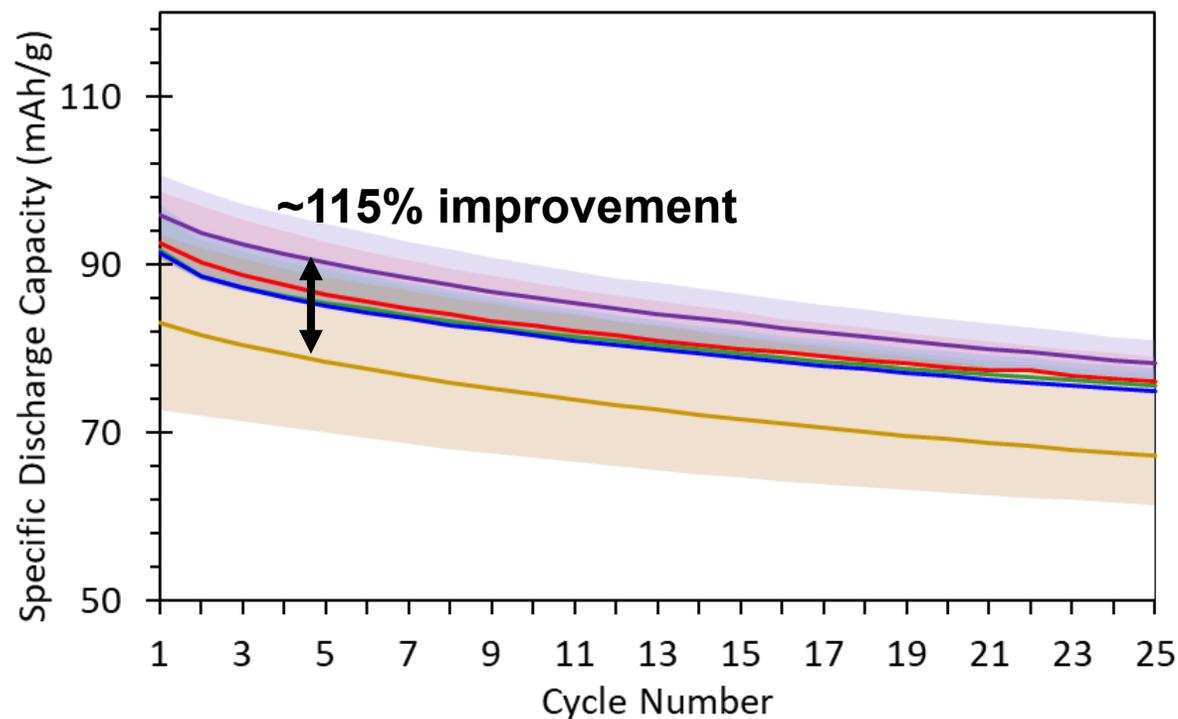


Data analysis conducted by Paul Gasper (NREL)



- KOH-only sample recovers pristine behavior with repeated electrochemical cycling; KOH+KCl sample shows continuous adverse reactivity
- Reaction in “non-NMC” electrochemical window: intercalation of K^+/Cl^- into graphite; LiOH oxidation

— Base Treatment Only (KOH) — Base Treatment + Various Post-Treatment Solvent Rinses

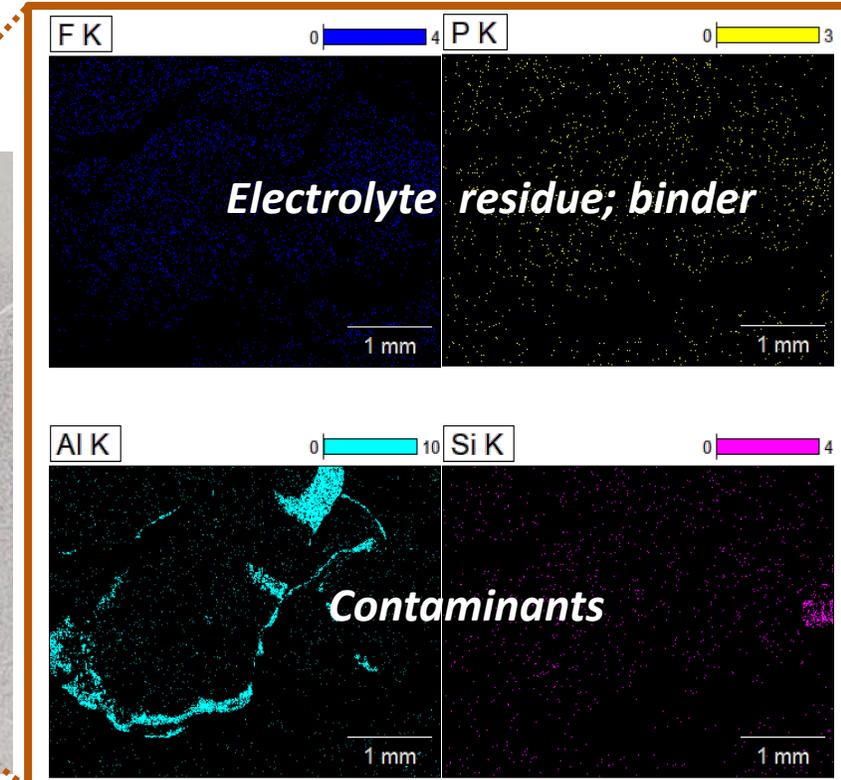
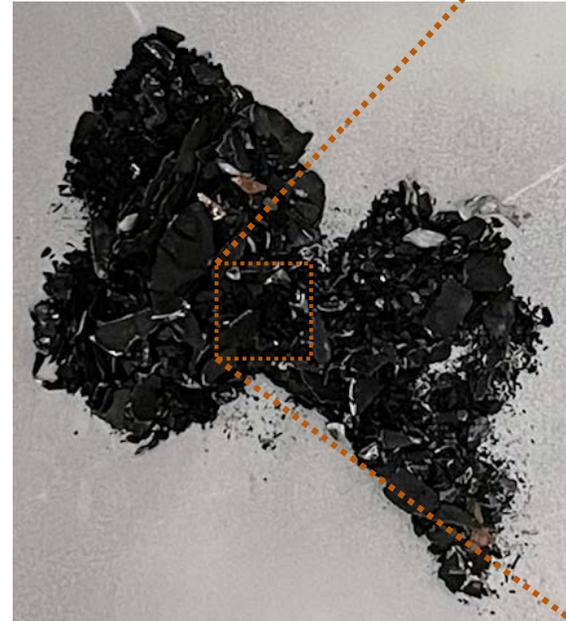
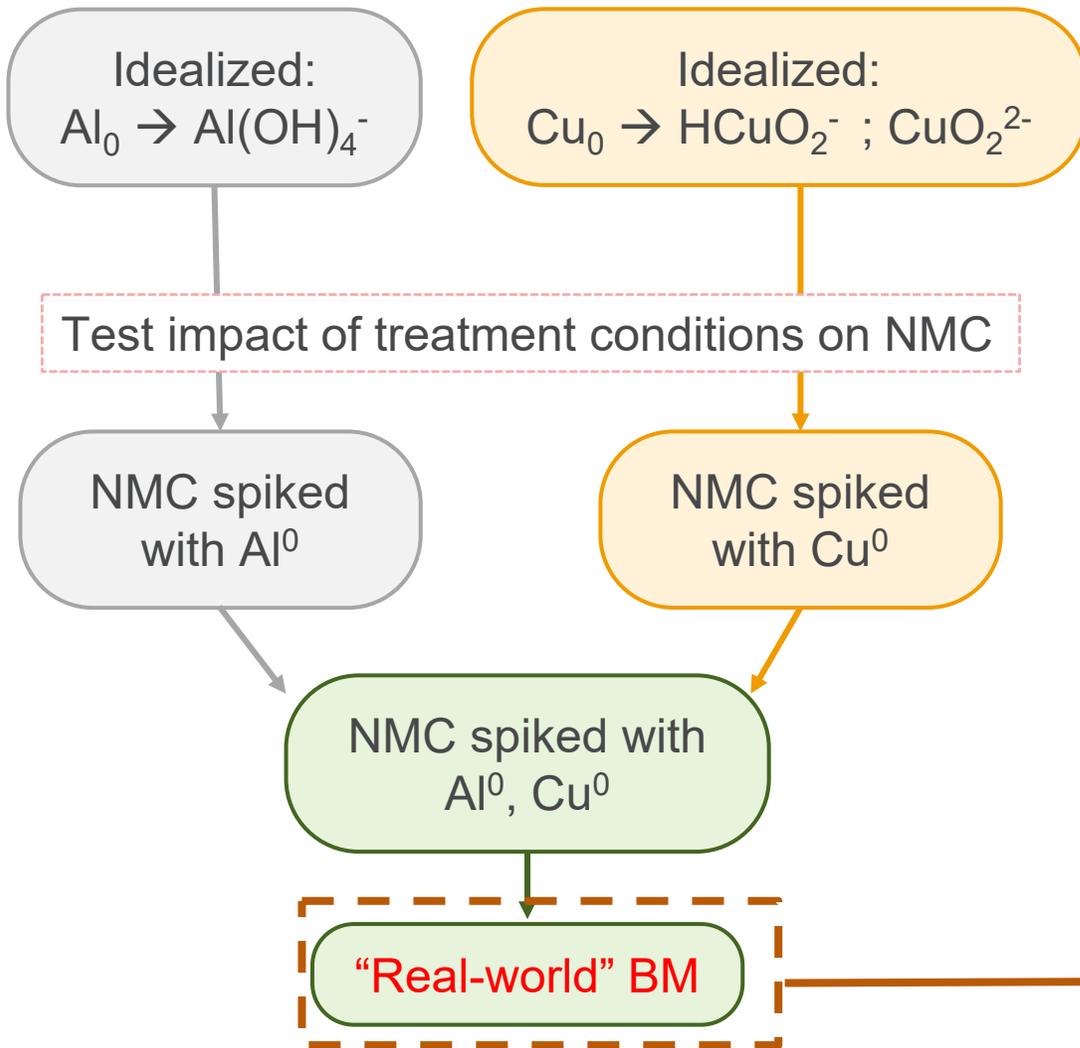


- Combining (electro)chemical analysis and chemical rationale, we attribute the reduced capacity observed in full cells primarily to the *use of DI water* as a post-treatment rinse solvent:
 - Surface-structural rearrangement
 - Ineffective removal of solubilized contaminants (Al) and treatment solvents (K)
- Evaluation of alternative post-treatment solvents is underway, with promising initial results for practical BM

Task #4: Treatment Efficacy on Practical BM



SEM/EDS imaging conducted
by Max Schulze (NREL)

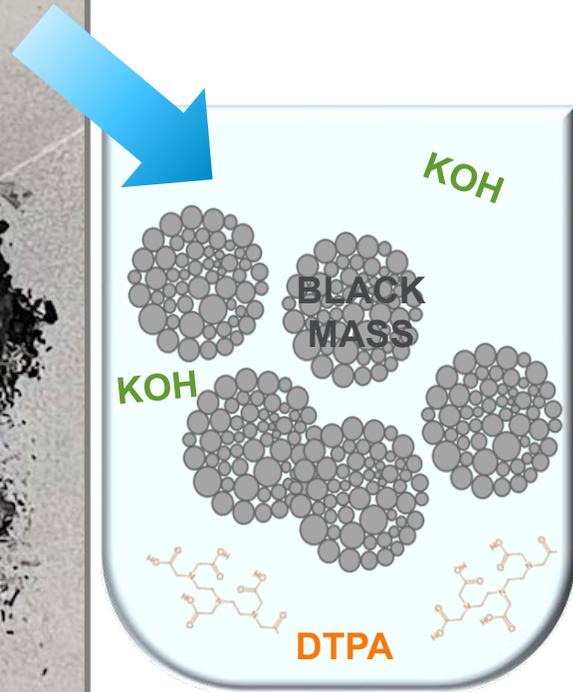


May contain additional metallic impurities not tested in idealized studies and/or non-NMC components (graphite, polymer separator, casing scraps, etc.)

Ongoing: Purification of (Practical) BM



- Larger shreds of NMC double-coated onto current collector
- Visible contaminant chunks
- Separator pieces present

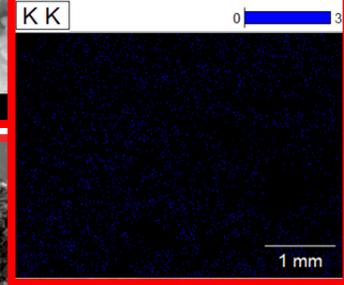
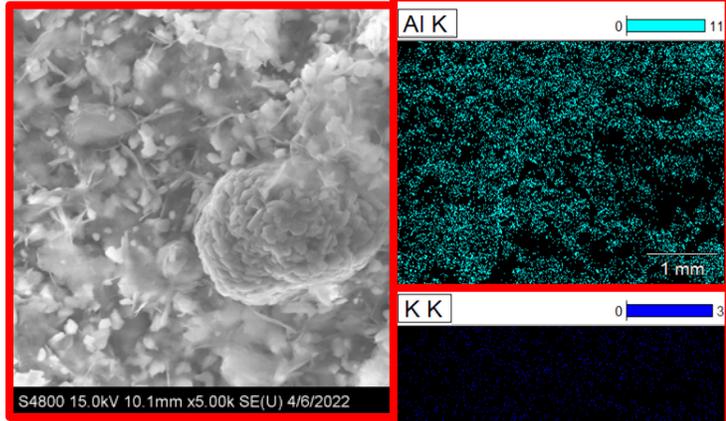


“BM Purification”:

- KOH → pH 13
- DTPA*
- 60 °C
- 2.5 hr stirring
- Sonication

*2x molar conc. of “worst-case” total est. contaminants (Cu + Mg + Fe) from fines (as determined via ICP)

Supernatant (recovered liquid phase) shows bluish color, indicating successful Cu ionization & binding



- Visible improvements to BM purity:
 - Dissolution of Al current collector
 - Major size reduction in Cu chunks
- However, some contamination still present (Cu particles; separator) and residual treatment species

Summary of Completed Work:

- ✓ Complete ionization of practically relevant concentrations of Al^0 , Cu^0 achieved under idealized conditions using low-cost, relatively mild processes
 - Sonication (Al^0 , Cu^0) and use of a strong chelating agent accelerate corrosion kinetics & reaction extent
- ✓ Cl^- salts found to *inhibit*, rather than enhance, Cu^0 corrosion at mildly elevated temperatures
- ✓ Salts with larger cationic radius (i.e., K^+) reduce Li substitution and improve performance of treated material
- ✓ Optimized purification process has successfully been demonstrated for NMC spiked with 1% Al^0 and 1% Cu^0 , with bulk structure and half-cell capacity of treated material matching pristine material
- ✓ Variability and reduced capacity in full-cells is attributed to surface changes (structure, chemistry) induced by post-treatment conditions (DI water washing) rather than the treatment process itself
 - Residual products are electrochemically consumed upon repeated cycling, restoring pristine behavior
- ✓ Demonstration of first application to industrial practical mass; optimization continues!

Ongoing Efforts:

- Optimized post-treatment conditions for applications to industrial BM (shredded end-of-life batteries)
- Scale up for demonstration at ReCell direct recycling pilot plant (current max: 15 g)
- Development of tailored sorbents for selective recovery of Al and Cu from solution

Thank you for your attention!

Further questions:
Kae.Fink@nrel.gov



www.recellcenter.org



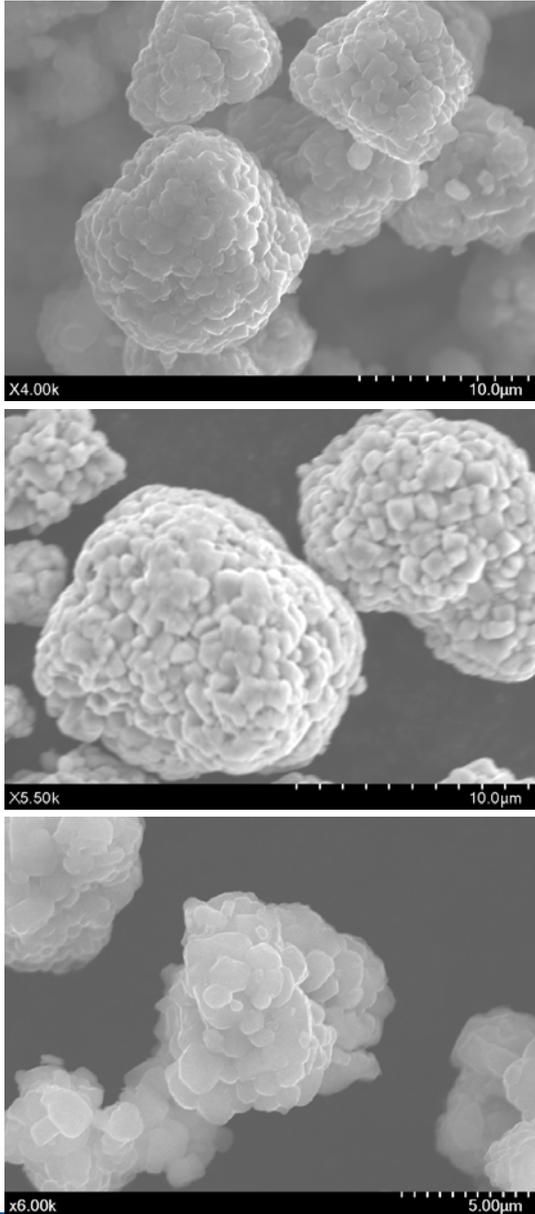
NREL/PR-5700-84258

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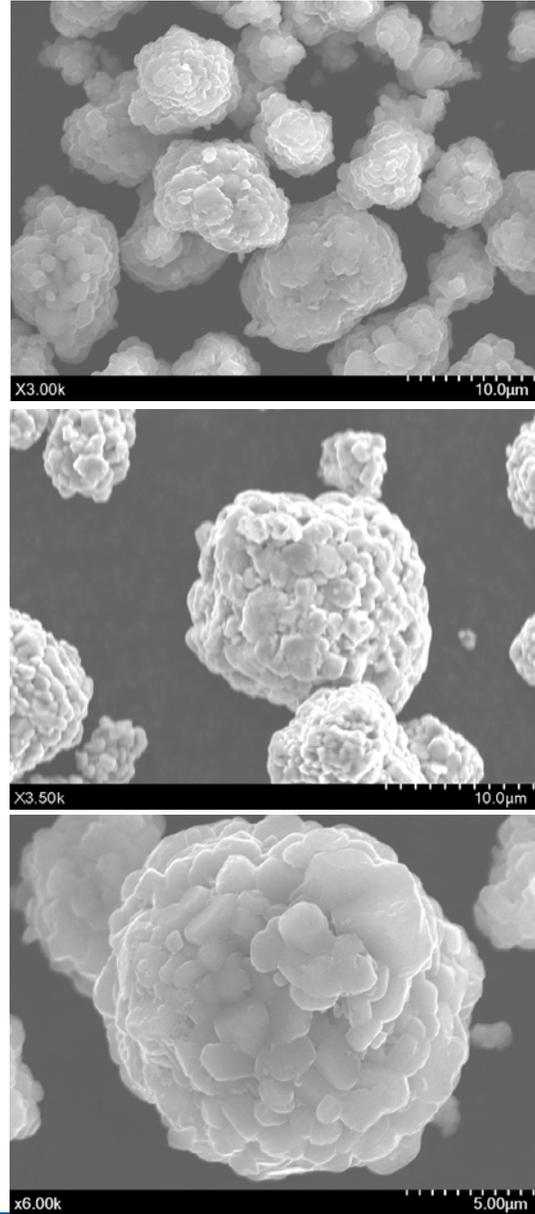


Morphological Impacts of Treatment on NMC

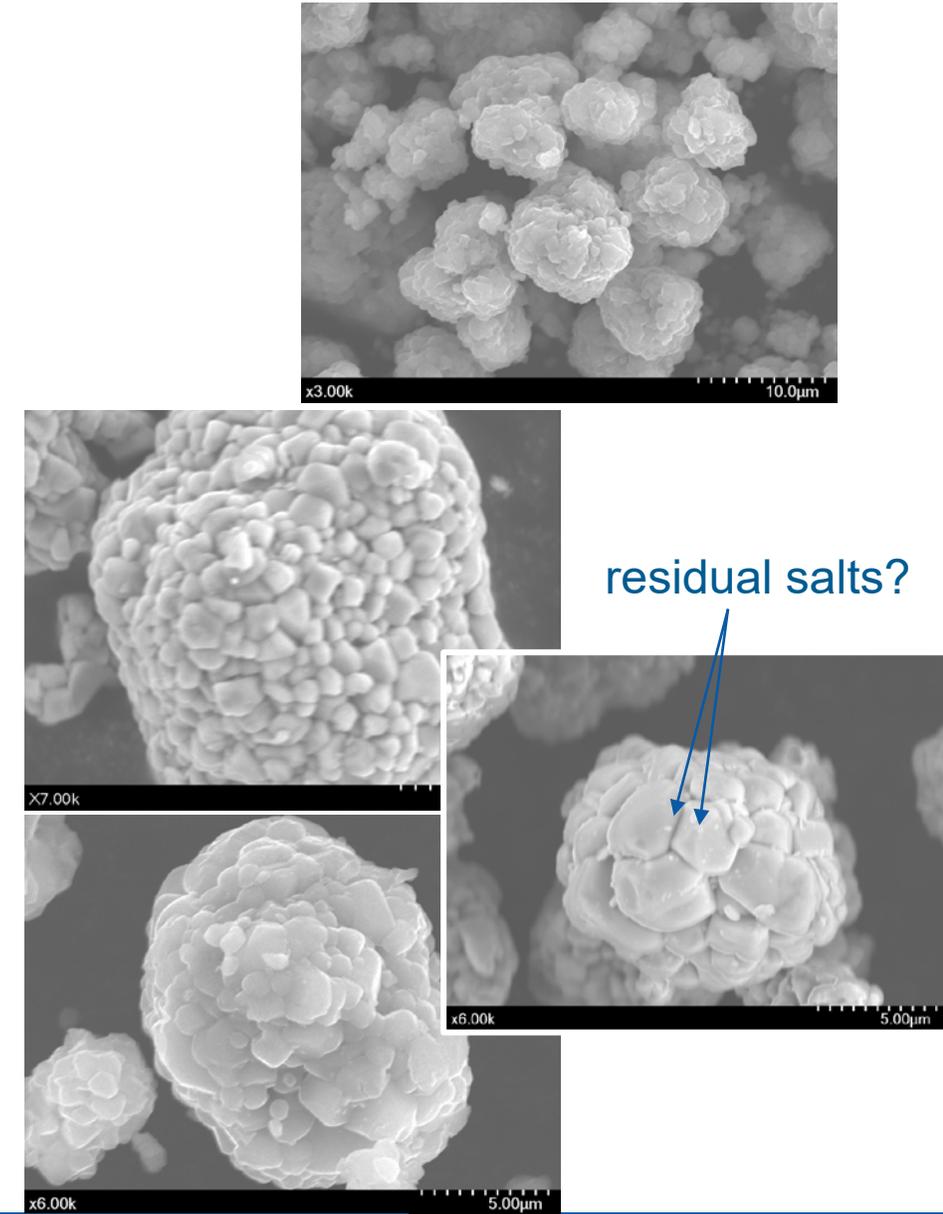
Pristine



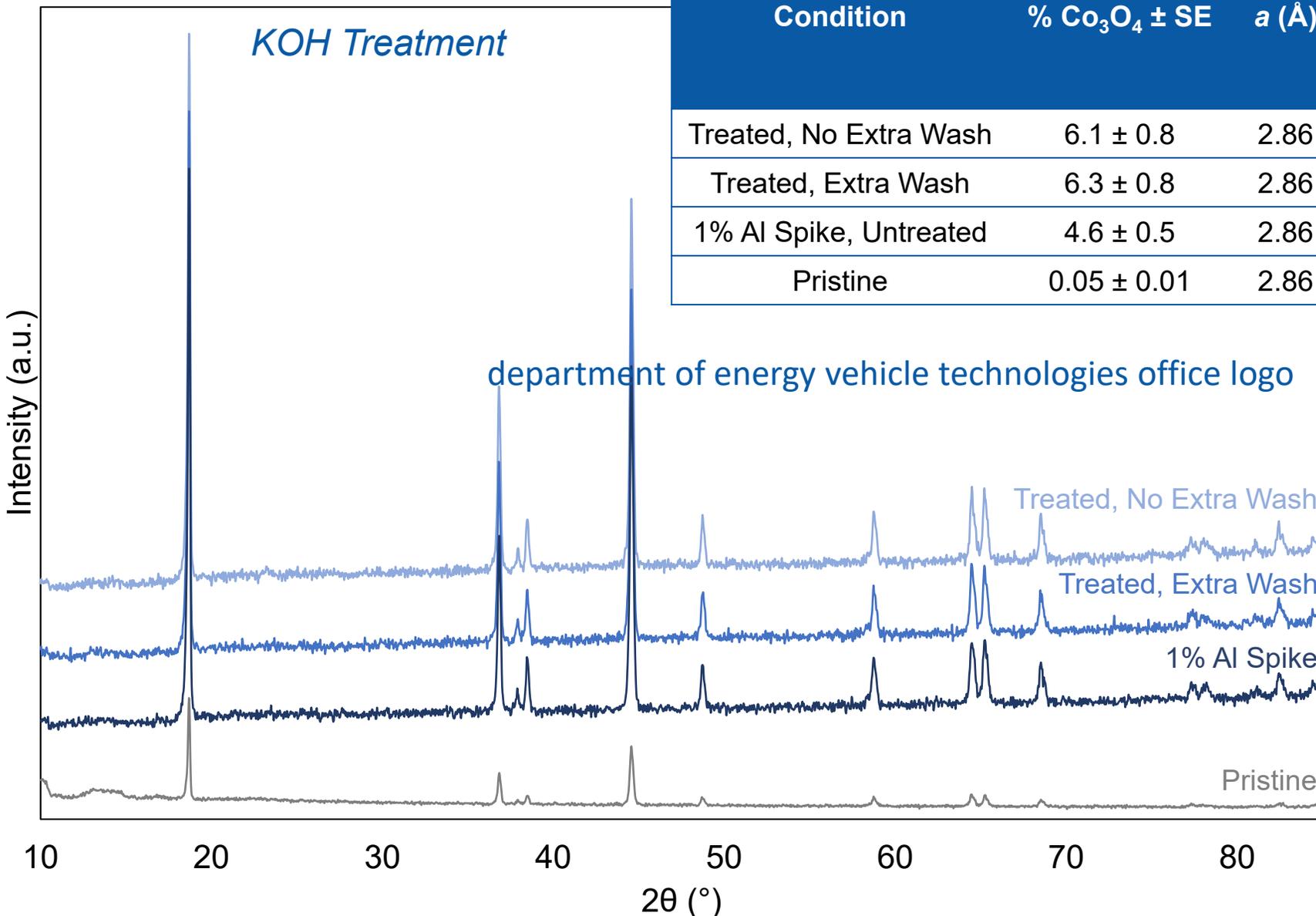
KOH Treatment



KOH + KCl Treatment



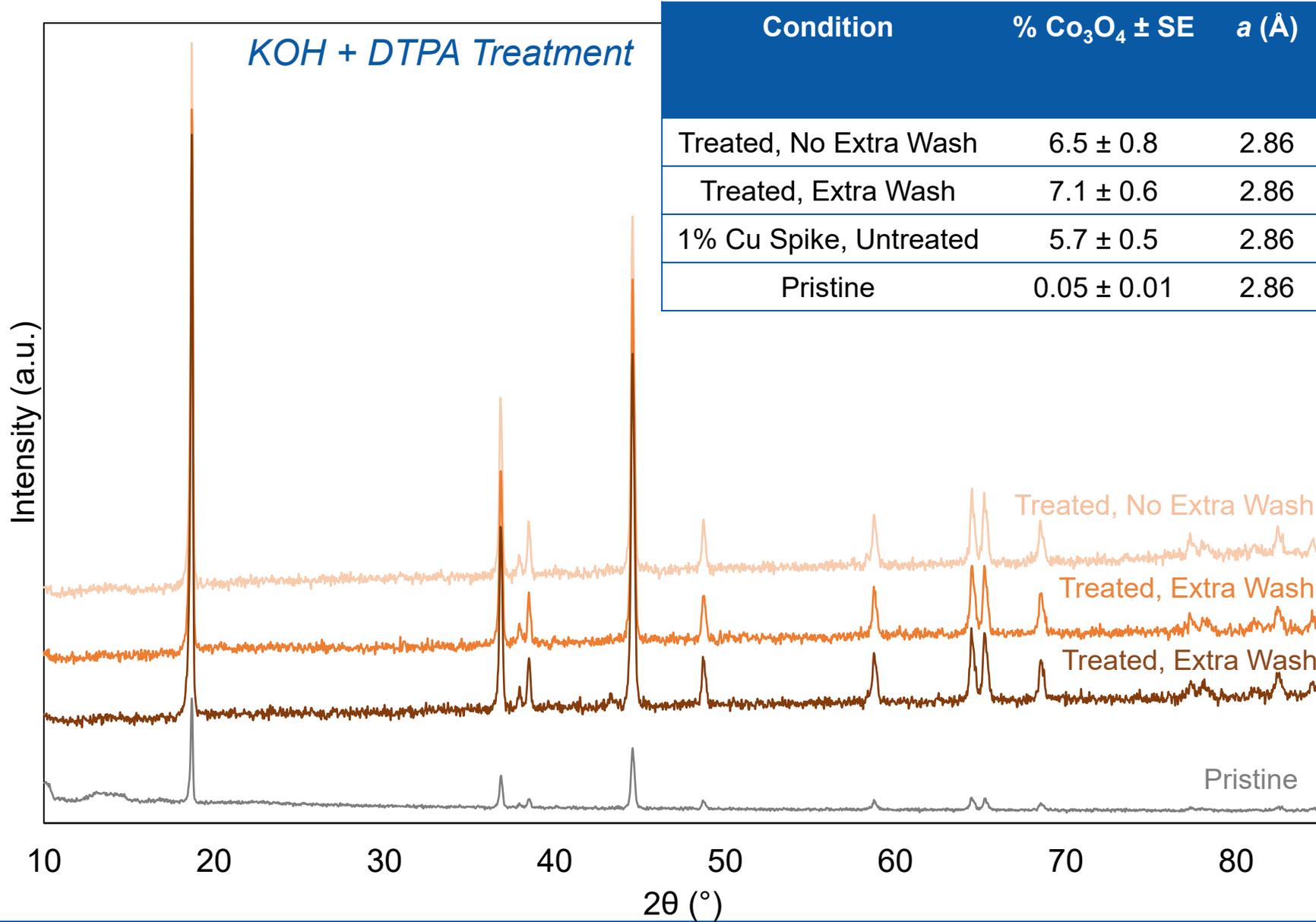
Structural Impacts of Treatment on NMC



Condition	% $\text{Co}_3\text{O}_4 \pm \text{SE}$	a (Å)	c (Å)	$I(003)/I(104)$	“R-factor”: $\{I(006)+I(012)\}/I(101)$
Treated, No Extra Wash	6.1 ± 0.8	2.86	14.23	1.19	0.360
Treated, Extra Wash	6.3 ± 0.8	2.86	14.23	1.20	0.401
1% Al Spike, Untreated	4.6 ± 0.5	2.86	14.23	1.20	0.426
Pristine	0.05 ± 0.01	2.86	14.23	1.05	0.379

Rietveld refinement suggests no impact of treatment on cation mixing or lattice parameters (i.e., no evidence of bulk Li loss, even with supplemental DI H₂O rinse). Some evidence of spinel phase transformation; impacts to long-term electrochemical stability are under investigation.

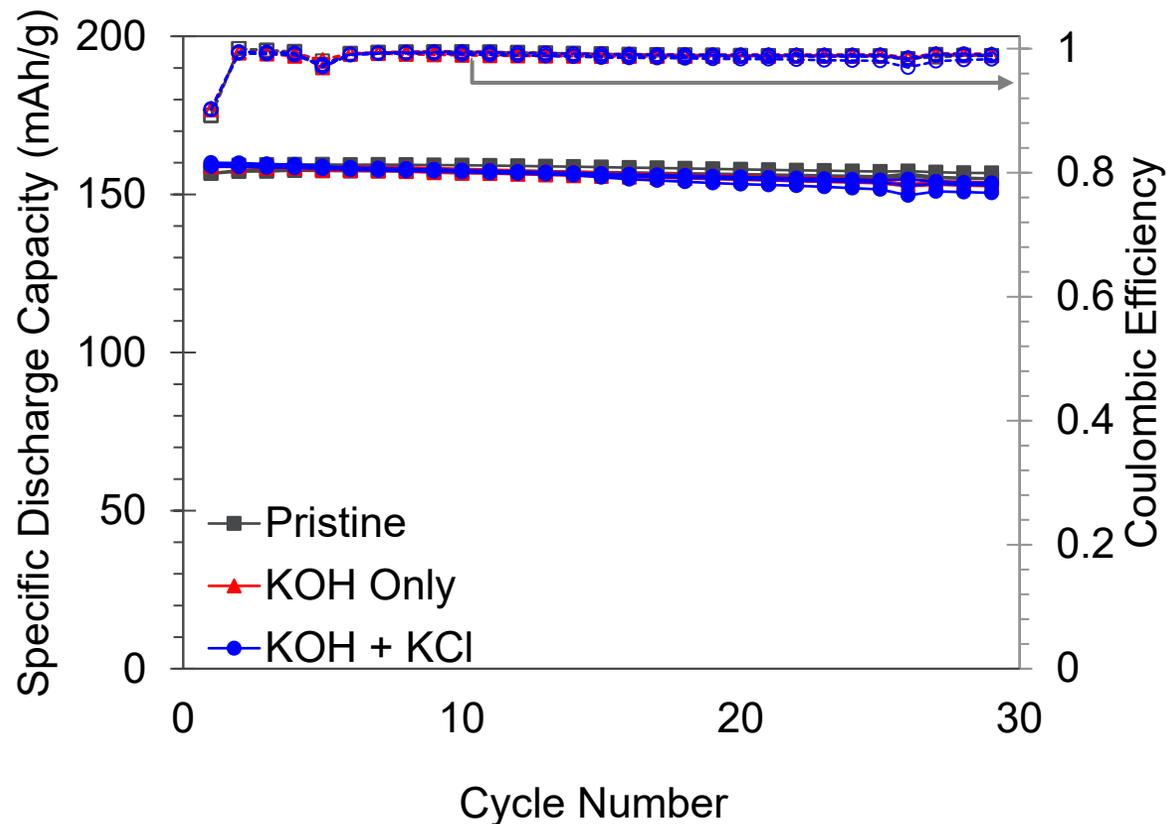
Structural Impacts of Treatment on NMC



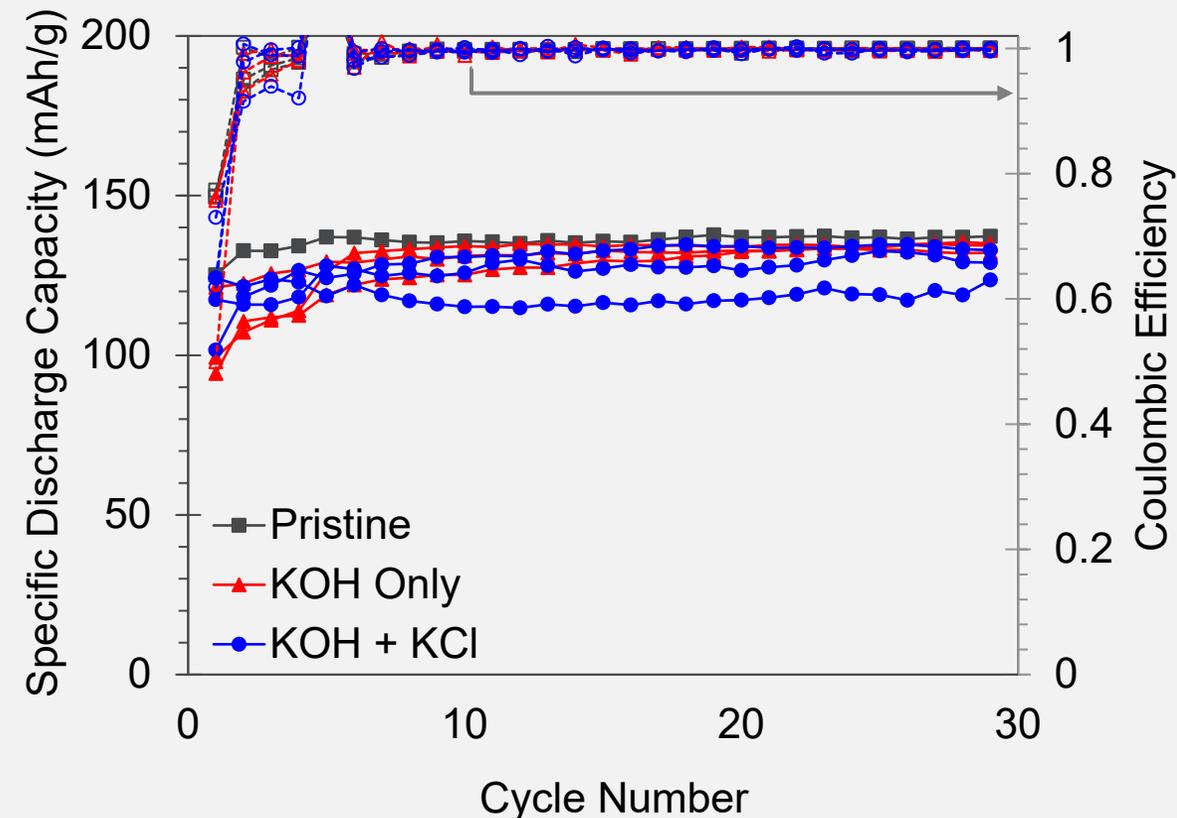
Condition	% $\text{Co}_3\text{O}_4 \pm \text{SE}$	a (Å)	c (Å)	$I(003)/I(104)$	“R-factor”: $\{I(006)+I(012)\}/I(101)$
Treated, No Extra Wash	6.5 ± 0.8	2.86	14.23	1.16	0.385
Treated, Extra Wash	7.1 ± 0.6	2.86	14.23	1.16	0.387
1% Cu Spike, Untreated	5.7 ± 0.5	2.86	14.23	1.20	0.378
Pristine	0.05 ± 0.01	2.86	14.23	1.05	0.379

No additional impact of DTPA presence on cation mixing or lattice parameters (i.e., no evidence of bulk Li loss caused by chelating agent).

Half Cells: NMC-111; K⁺ Salts

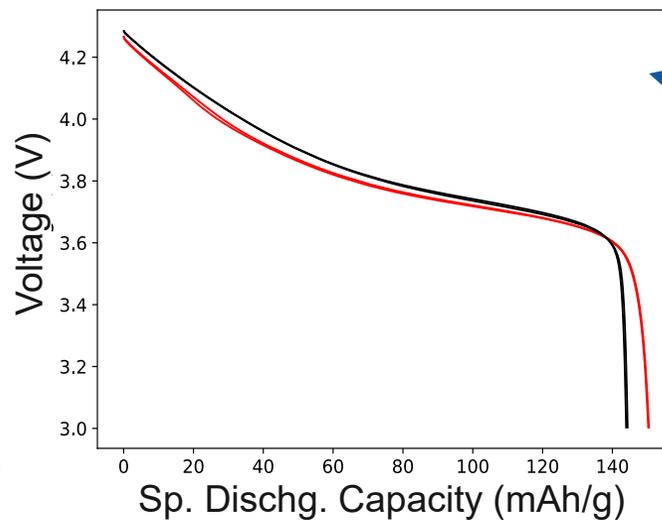
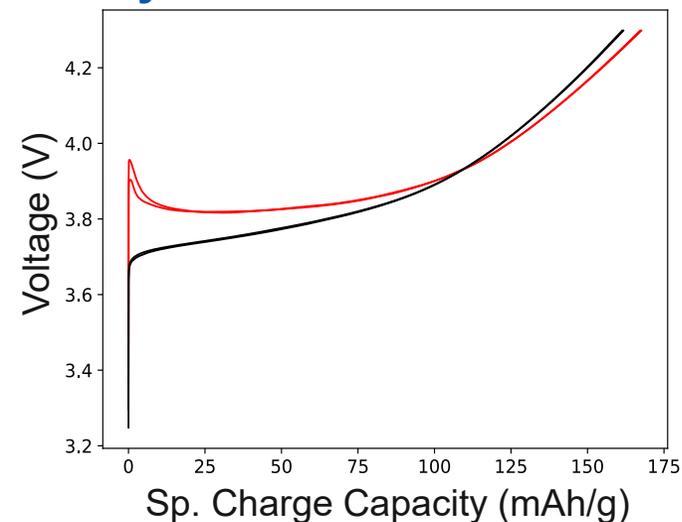


Full Cells: NMC-111; K⁺ Salts (n/p: ~1.15)

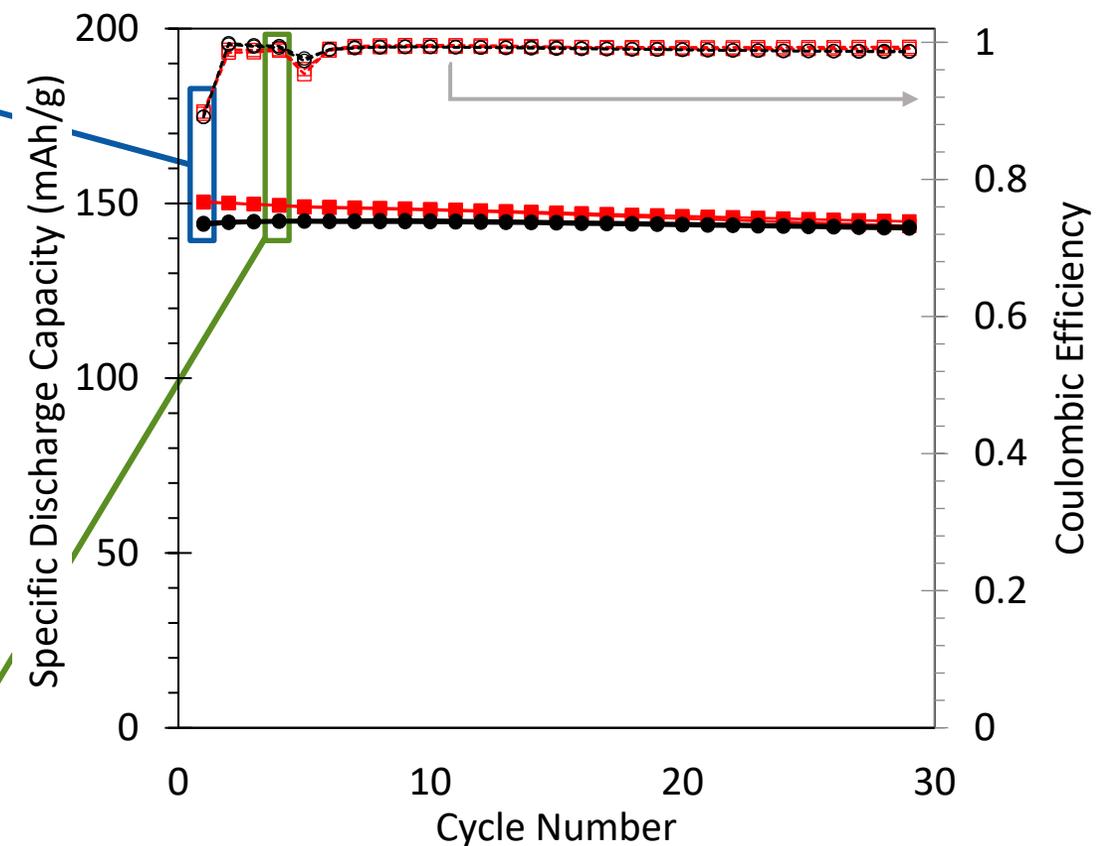
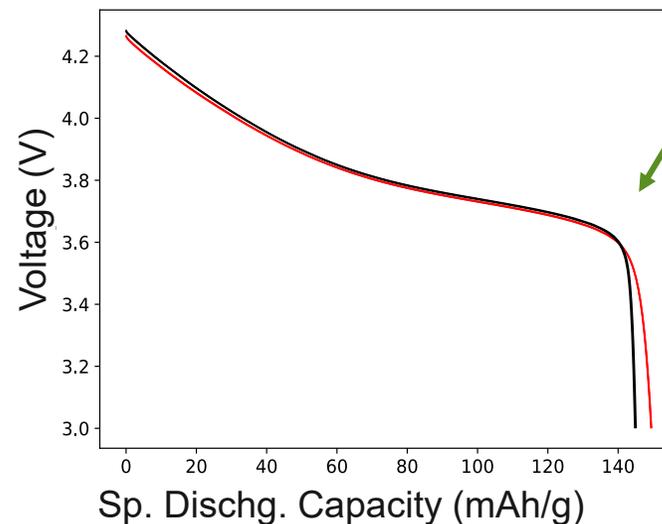
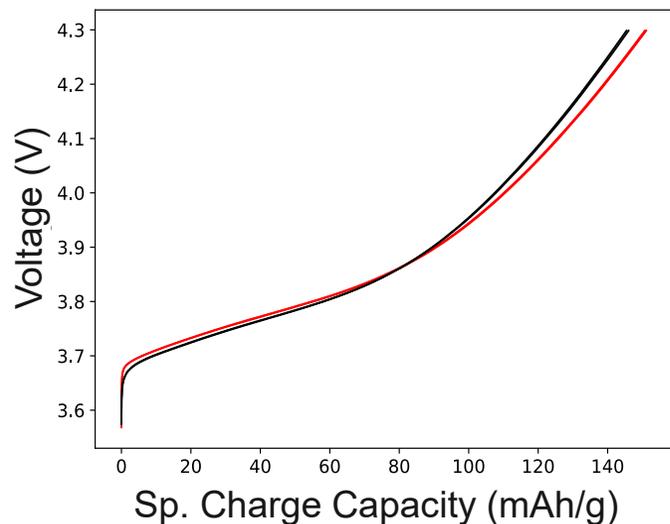


- NMC treated with K⁺ salts shows improved performance over NMC treated with Na⁺ salts (reduced cationic substitution)
- Reduced capacity in full cells may be attributable to residual Cl⁻ salts on the surface of NMC (irregular SEI formation).
- Mitigation strategies for residual salts under investigation; simple DI H₂O wash may leach Li⁺

Cycle 1



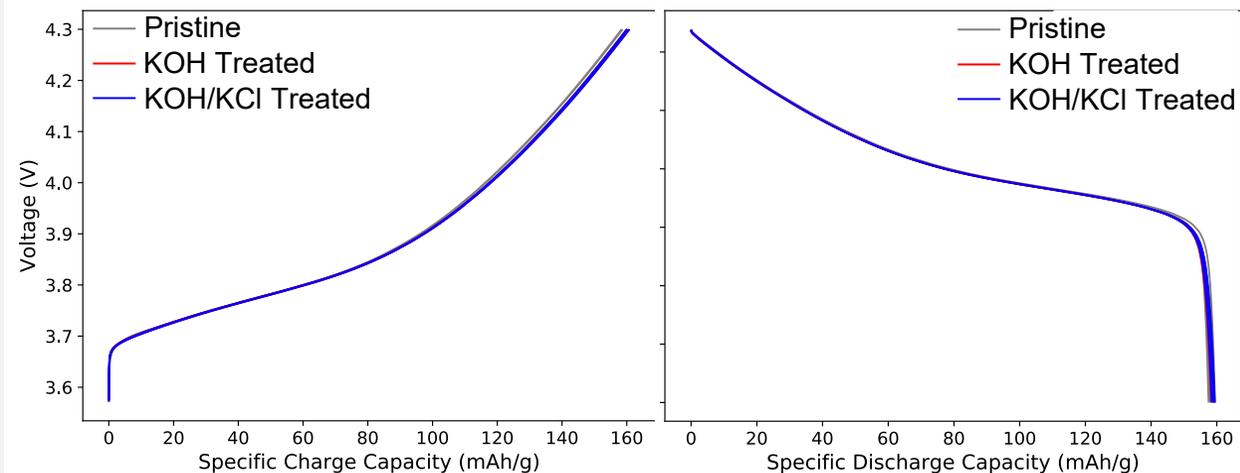
Cycle 4



- Irregular first-cycle behavior may be attributable to residual salts on the surface of NMC (affecting SEI evolution). Performance stabilizes after formation.
- Mitigation strategies for residual salts under investigation; simple DI H₂O wash may leach Li⁺.

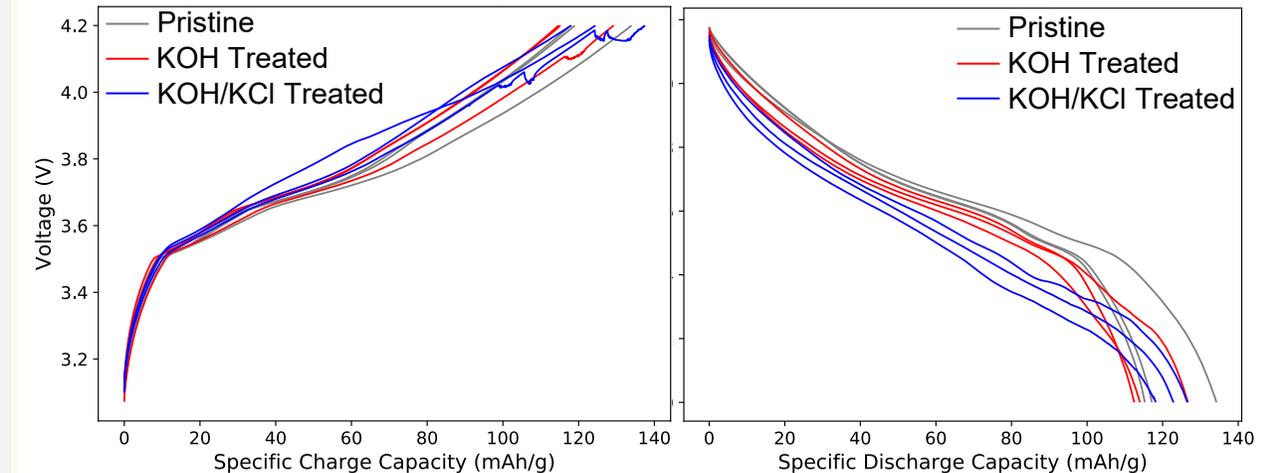
Half Cells: NMC-111; K⁺ Salts

Cycle 4 (end of formation)

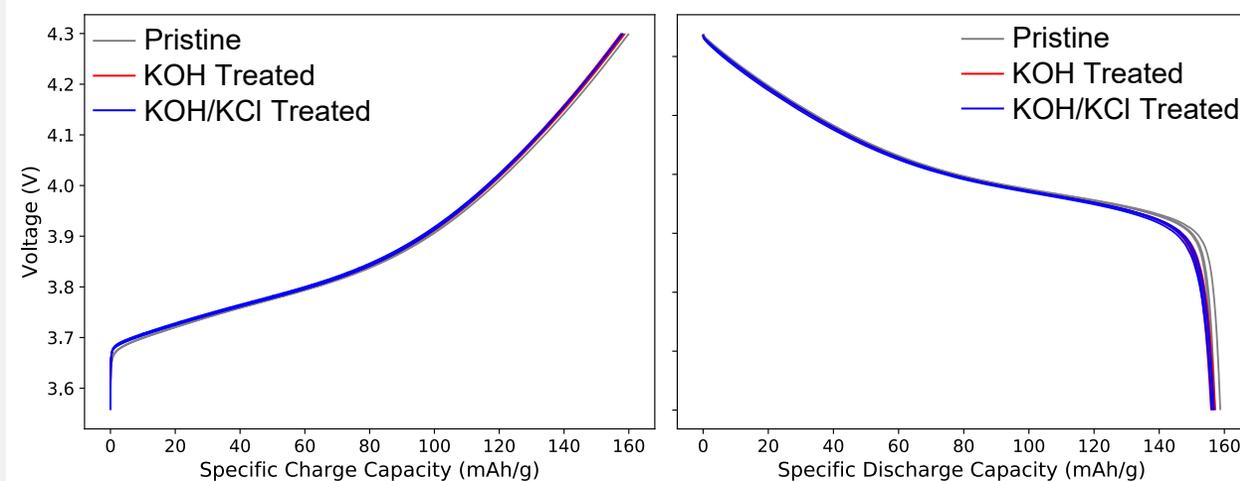


Full Cells: NMC-111; K⁺ Salts

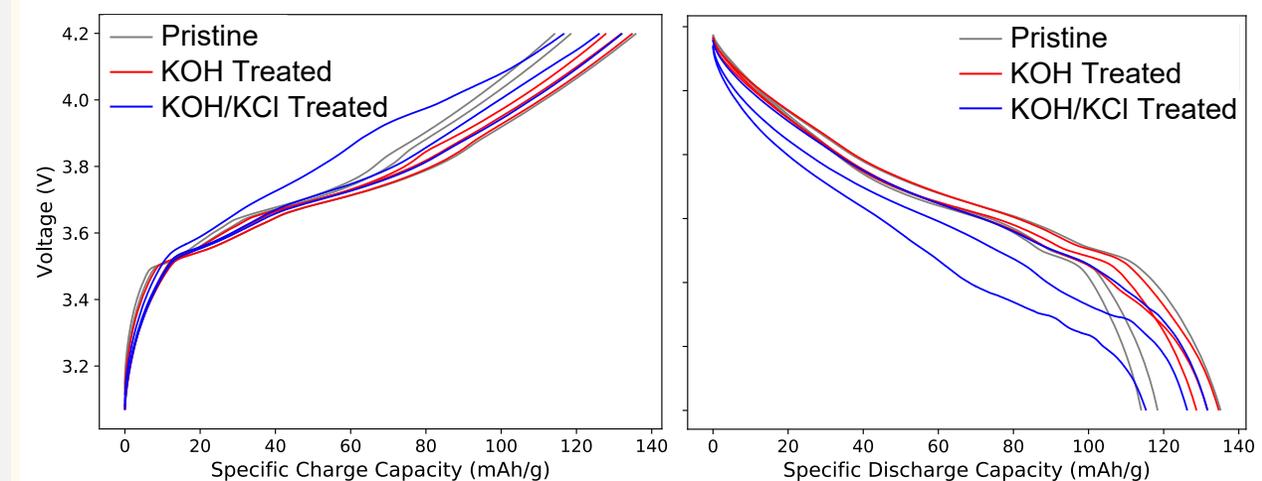
Cycle 4 (end of formation)



Cycle 14 (10 cycles @ C/10)



Cycle 14 (10 cycles @ C/10)



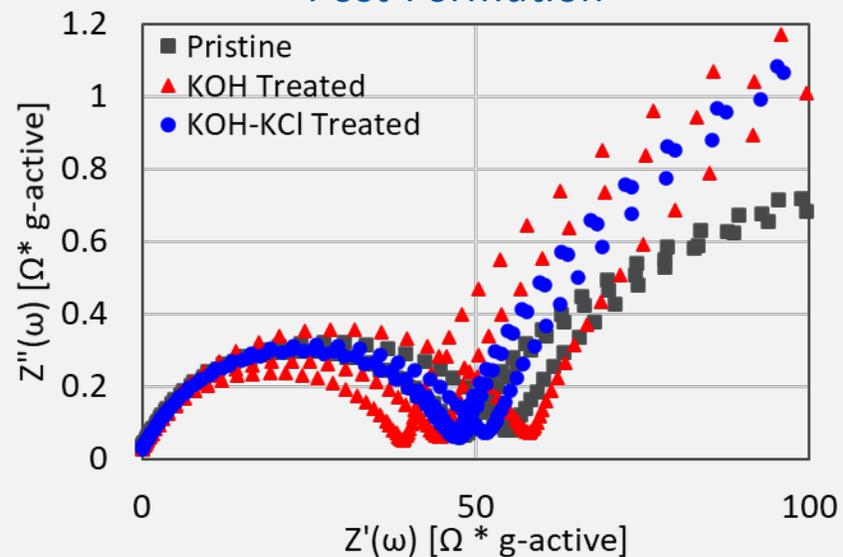
Change in voltage profiles for treated full cells is thought to be from residual salts, and not cathode degradation.

IMPEDANCE DATA (EIS); TREATED NMC-111 & NMC-532; POST-FORM.

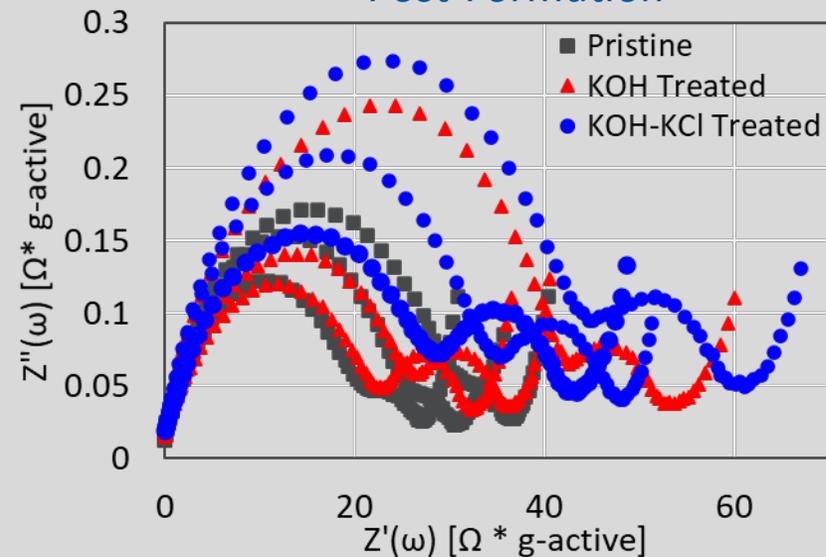
Half Cells: NMC-111; K⁺ Salts

Full Cells: NMC-111; K⁺ Salts (n/p: ~1.15)

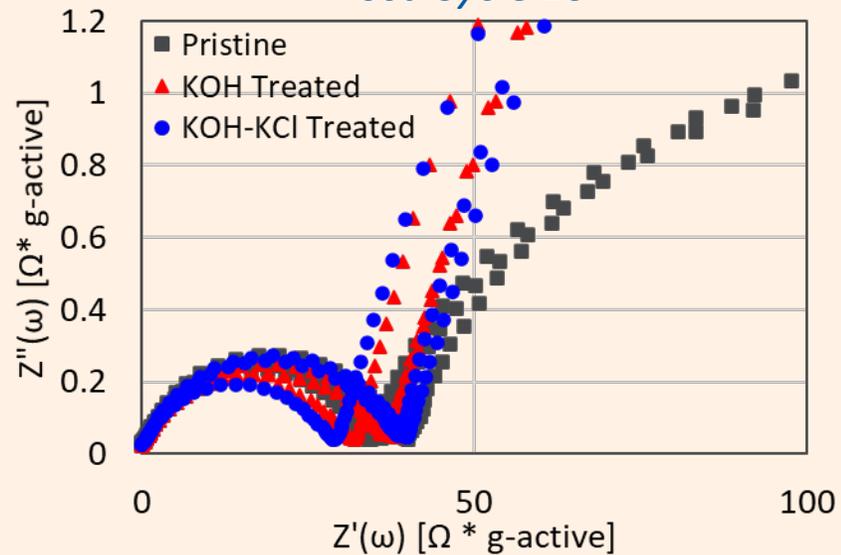
Post-Formation



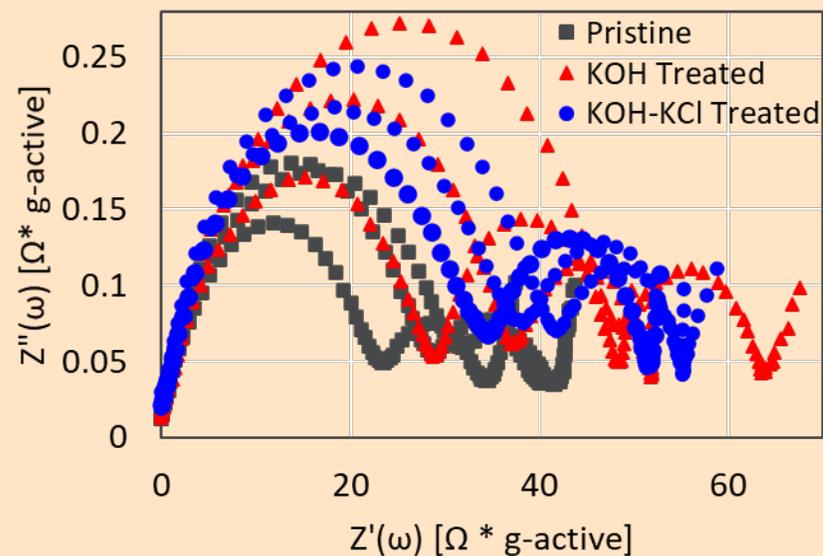
Post-Formation



Post-Cycle 25

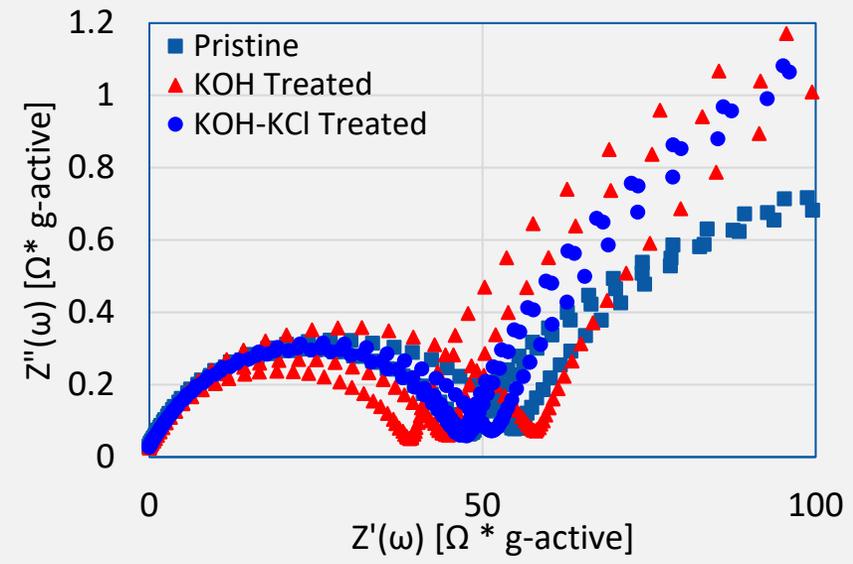


Post-Cycle 25

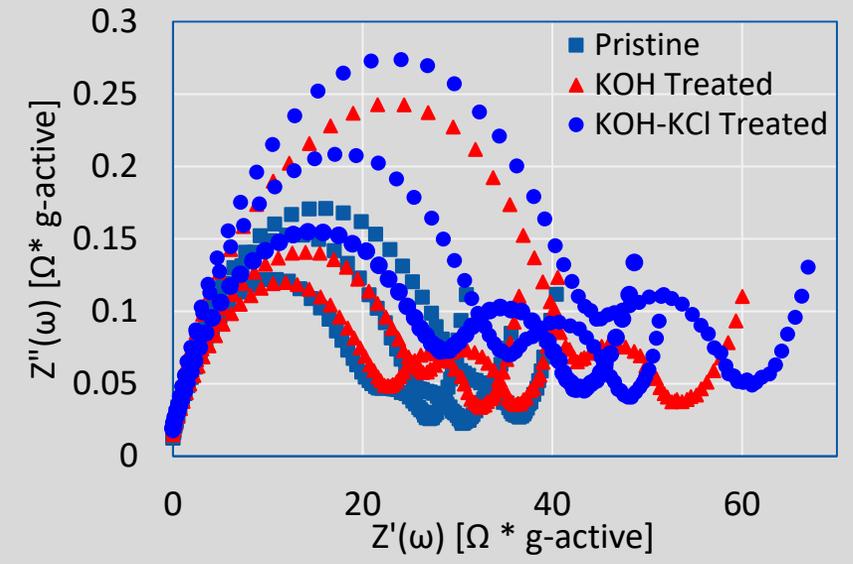


IMPEDANCE DATA (EIS); TREATED NMC-111 & NMC-532; POST-FORM.

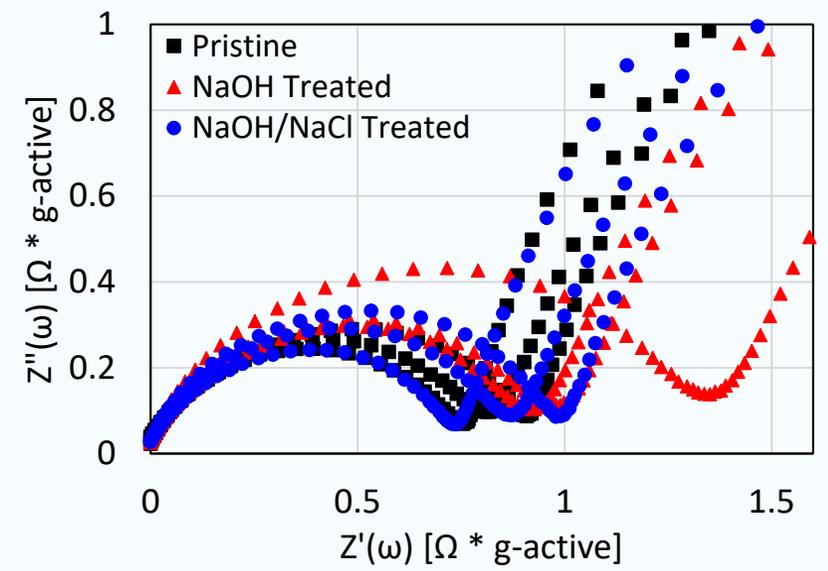
Half Cells: NMC-111; K⁺ Salts



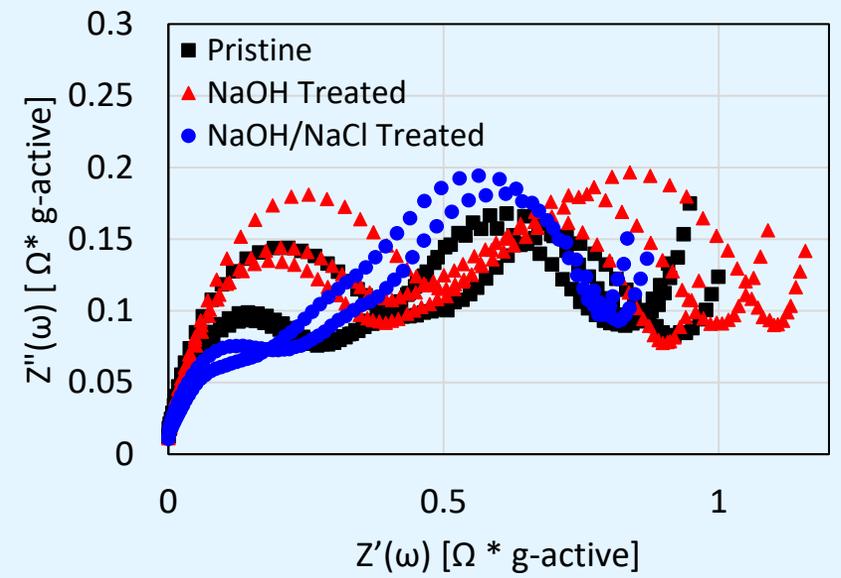
Full Cells: NMC-111; K⁺ Salts (n/p: ~1.15)



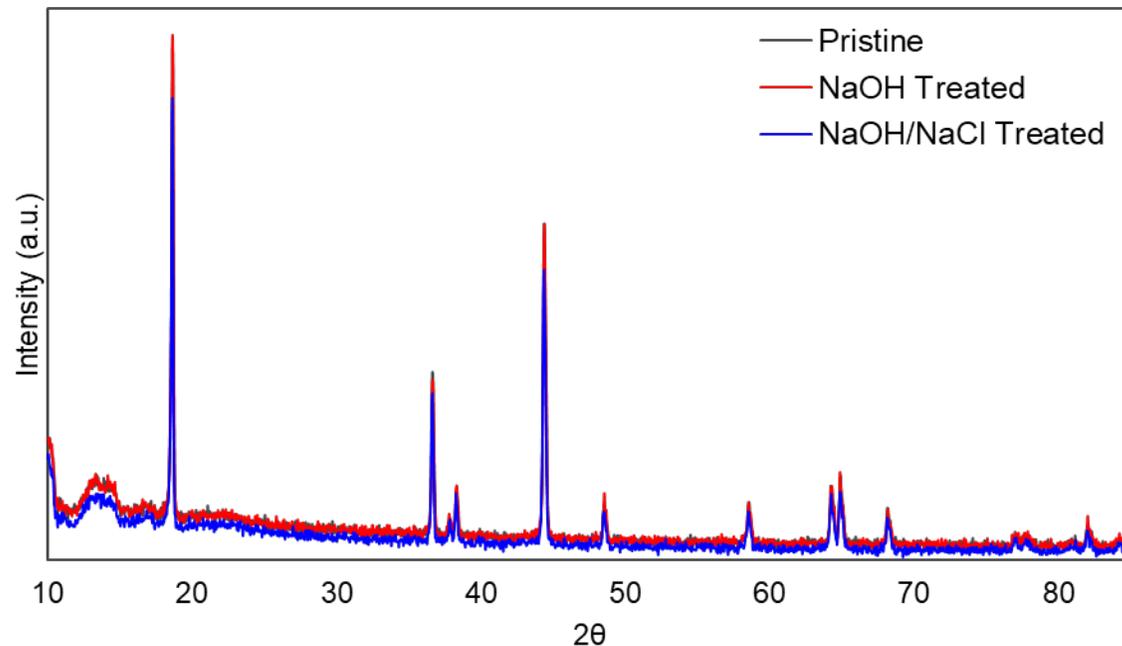
Half Cells: NMC-532; Na⁺ Salts



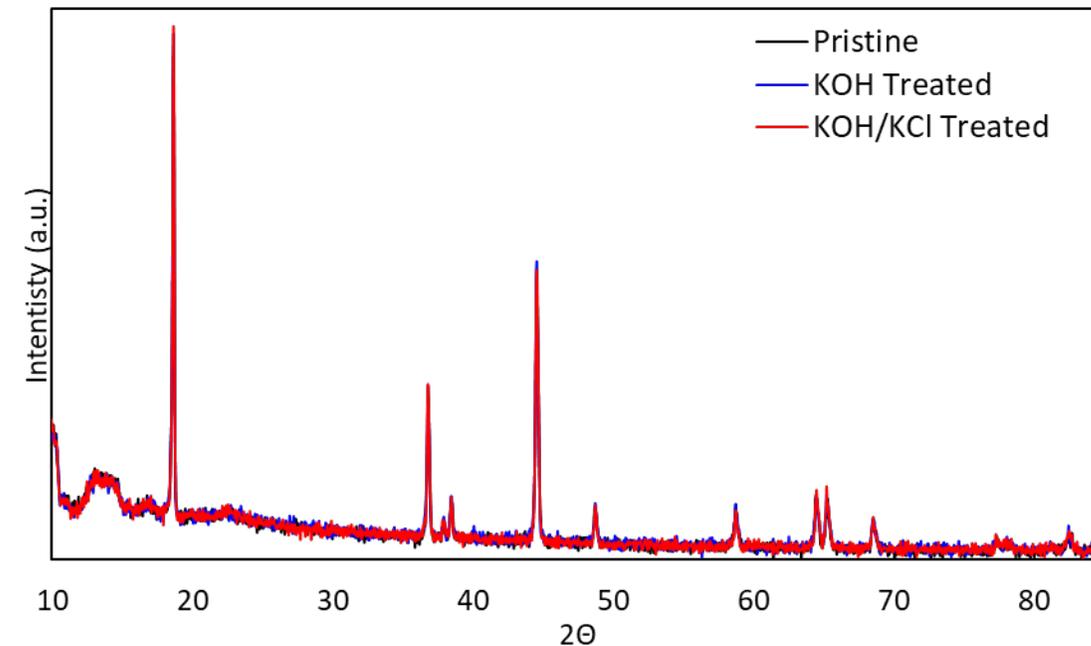
Full Cells: NMC-532; Na⁺ Salts (n/p: ~1.25)



NMC-532; Na⁺ salts; pH 11



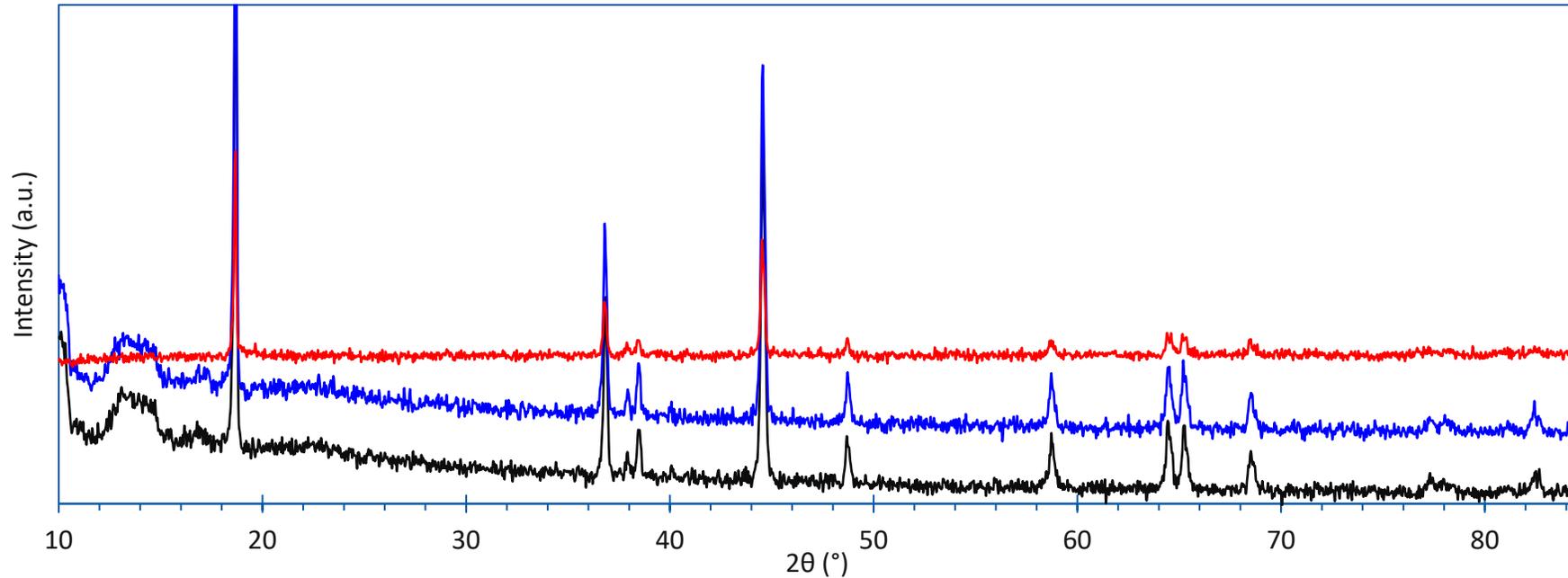
NMC-111; K⁺ salts; pH 13



Condition	% Co ₃ O ₄ ± SE	<i>a</i>	<i>c</i>	<i>I</i> (003)/ <i>I</i> (104)
Pristine	1.0 ± 0.7	0.288	1.426	1.03
NaOH	1.5 ± 0.9	0.289	1.425	1.06
NaOH/NaCl	1.1 ± 0.5	0.288	1.426	1.12

Condition	% Co ₃ O ₄ ± SE	<i>a</i>	<i>c</i>	<i>I</i> (003)/ <i>I</i> (104)
Pristine	0.05 ± 0.01	0.286	1.423	1.05
KOH	0.04 ± 0.01	0.286	1.423	1.08
KOH/KCl	0.05 ± 0.01	0.286	1.422	1.09

No significant bulk structural transformations (cation mixing, lattice expansion, phase change) observed for treated NMC.



Impacts of Optimized Treatment Conditions on NMC Structure:

- No observed change to bulk lattice parameters; no significant peak shifting
- Conflicting evidence around cation mixing...may need to repeat sample due to low intensity
- Possible increase in spinel phase (within range of refinement error)

NMC	Treatment Type	pH	Temperature	Agitation	Sonics?	Time (hr)	% $\text{Co}_3\text{O}_4 \pm \text{SE}$	<i>a</i>	<i>c</i>	<i>I</i> (003)/ <i>I</i> (104)	"R-factor": $\{I(006)+I(012)\}/I(101)$
532	Pristine	---	---	---	---	---	1.0 ± 0.7	0.288	1.426	1.03	0.408
532	NaOH	11	Ambient	Shaking	No	1	1.5 ± 0.9	0.289	1.425	1.06	0.458
532	NaOH/NaCl	11	Ambient	Shaking	No	1	1.1 ± 0.5	0.288	1.426	1.12	0.443
* 111	Pristine	---	---	---	---	---	0.05 ± 0.01	0.286	1.423	1.05	0.379
* 111	KOH	13	Ambient	Shaking	No	1	0.04 ± 0.01	0.286	1.423	1.08	0.376
111	KOH/KCl	13	Ambient	Shaking	No	1	0.05 ± 0.01	0.286	1.422	1.09	0.333
* 111	KOH	13	60 °C	Overhead Stir	Yes	2	0.07 ± 0.01	0.286	1.422	1.18	0.505

*Reduced intensity of red sample due to smaller divergence slit (low sample volume); repeat analysis planned